may in fact determine the conformations of particular residues through significant dipole–dipole interactions. The purpose of presenting the results of Table III here is merely to demonstrate that the approximate method is a valid one. Evaluation of the actual role played by long-range electrostatic effects in such processes as protein folding remains to be performed.

The computational time savings using the dipole method for the long-range interactions in a protein the size of pancreatic trypsin inhibitor is about 2.5. Thus, it also results in substantial reductions in computational times.

Acknowledgment. We are indebted to Drs. L. G. Dunfield and G. Nemethy for helpful comments on this manuscript.

References and Notes

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(2) (a) NIH Postdoctoral Fellow, 1975–1976. (b) To whom requests for reprints should be addressed.


(5) For a recent discussion of empirical algorithms, see F. R. Maxfield and H. A. Scheraga, Biochemistry, 15, 5138 (1976).


(20) R. Huber, personal communication.


Effect of Gallium Ions and of Preparation Methods on the Structural Properties of Cobalt–Molybdenum–Alumina Catalysts

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Small amounts of gallium ions were added to γ-alumina and their influence on the structural properties of the system Co–(Mo)/γ-Al2O3 was studied. It is shown that, due to the presence of Ga3+ ions, a “surface spinel” (MnAl2O4) with a larger amount of Co2+ in tetrahedral sites as compared to the spinel formed on CoA12O4 is formed with a larger amount of Co2+ in tetrahedral sites as compared to the spinel formed on CoA12O4 is formed. A decrease of the segregated CoO4 is also observed. A possible effect of gallium ions on molybdenum is discussed. It is also reported that different preparation methods (single or double impregnation) lead to the formation of different surface species. Cobalt aluminate, molybdate monolayer, and CoO4, depending on the Co content, are formed on doubly impregnated specimens. Cobalt aluminate and cobalt molybdate are the main species formed on singly impregnated specimens. Finally brief consideration is given to how the Co and Mo species, present in the oxide form, change in the sulfided form.

1. Introduction

In previous studies of supported oxide systems on alumina, it was shown that a “surface spinel” (MnAl2O4, CoAl2O4, NiAl2O4, CuAl2O4) was formed, either alone or in addition to an oxide phase. Additions of trace amounts of Zn2+, Ga3+, and Ge4+ ions modify the surface properties of alumina, thus affecting the structural features of supported transition metal ions. The presence of these ions, all having a preference for the tetrahedral site, favors a normal cation distribution in the surface spinel NiAl2O4. Since the type of symmetry adopted by supported cobalt and molybdenum directly influences their reactivity, it was of interest to investigate the influence of Ga3+ ions in the hydrodesulfurization (HDS) of Co–Mo–alumina catalysts. Within this framework we have also examined how the order of addition of transition metal ion promoters affects the structural properties of the Co–Mo/Ga2O3–γ-Al2O3 system in the oxide form, and how the Co and Mo surface species present in the oxide forms are related to those developed in the sulfided catalysts.

2. Experimental Section

2.1. Catalyst Preparation. The gallium-containing γ-alfaumina support (AyGa) was prepared by impregnating γ-Al2O3 with gallium nitrate. The soaked mass, dried at 500 °C for 15 h, was used as a support for the CoMo catalyst. Portions of the γ-Al2O3 and γ-Al2O3 supports were impregnated with a solution of cobalt nitrate of hydrodesulfurization (HDS) of Co–Mo–alumina catalysts. Within this framework we have also examined how the order of addition of transition metal ion promoters affects the structural properties of the Co–Mo/Ga2O3–γ-Al2O3 system in the oxide form, and how the Co and Mo surface species present in the oxide forms are related to those developed in the sulfided catalysts.

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The supports AyGa0.6 and AyGa4 were impregnated with an ammonium heptamolybdate solution in the same amount as in method A followed only by drying at 120 °C. Different portions of this material were again impregnated with a solution of cobalt nitrate. The mass was then dried at 120 °C, ground, and fired at 600 °C, ground, and fired at 120 °C, again impregnated with a solution of cobalt nitrate. The mass was then dried at 120 °C, ground, and fired at 600 °C, ground, and fired at 600 °C for 24 h. The catalysts so obtained are designated as A-AyGaMoCo(0.6:5:x) and A-AyGaMoCo(4:5:x) where x has the same values as before.

**Sulfurization.** Portions of AyGaMoCo(0.6:5:x) and AyGaMoCo(0.6:5:x) catalysts were sulfided in a silica reactor at 400 °C, for 2 h, in a flow of H₂S (90 cm³/min) and H₂S (10 cm³/min).

2.2. Physical Characterization and Chemical Analysis. Optical reflectance spectra were recorded on a Beckman DK 1 instrument, in the range 2500–210 nm at room temperature, using γ-Al₂O₃ as a reference. To check the influence of grain size on the reflectance spectra, some specimens were ground for 5 and 20 h in a mechanical mortar. The same spectrum was obtained in both cases. Therefore all spectra were recorded for samples ground for 5 h.

Magnetic susceptibility measurements were carried out by the Gouy method in the temperature range 100–295 K. The specimens were contained in a sealed silica tube. X-ray analysis was carried out with Co radiation, using a Debye–Scherrer camera (114 nm diameter) or a diffractometer (Phillips).

Chemical analysis for cobalt was performed by atomic absorption techniques (Varian Techtron AA5); concentration. The material was then dried at 120 °C, ground, and fired at 600 °C in air for 24 h. The catalysts so obtained are designated as B-AyGaMoCo(0.6:5:x) and B-AyGaMoCo(4:5:x) where x has the same values as before.

### Table I: Gallium-Containing Specimens and Their Properties

<table>
<thead>
<tr>
<th>Samples</th>
<th>Co content,a</th>
<th>Curie constant, C, erg G⁻¹ mol⁻¹ K</th>
<th>Magnetic moment, μB</th>
<th>Weiss temp -ε, K</th>
<th>Co²⁺ tot.%</th>
</tr>
</thead>
<tbody>
<tr>
<td>AyGaCo(0.6:1)</td>
<td>1.12</td>
<td>3.11</td>
<td>4.99</td>
<td>20</td>
<td>42</td>
</tr>
<tr>
<td>AyGaCo(0.6:2)</td>
<td>2.29</td>
<td>2.95</td>
<td>4.86</td>
<td>25</td>
<td>59</td>
</tr>
<tr>
<td>AyGaCo(0.6:3)</td>
<td>3.15</td>
<td>2.96</td>
<td>4.87</td>
<td>35</td>
<td>58</td>
</tr>
<tr>
<td>AyGaCo(0.6:4)</td>
<td>4.32</td>
<td>2.96</td>
<td>4.87</td>
<td>38</td>
<td>58</td>
</tr>
<tr>
<td>AyGaCo(0.6:5)</td>
<td>5.29</td>
<td>2.95</td>
<td>4.86</td>
<td>46</td>
<td>58</td>
</tr>
<tr>
<td>A-AyGaMoCo(0.6:5:1)</td>
<td>1.11</td>
<td>2.97</td>
<td>4.80</td>
<td>25</td>
<td>57</td>
</tr>
<tr>
<td>A-AyGaMoCo(0.6:5:2)</td>
<td>2.11</td>
<td>2.97</td>
<td>4.90</td>
<td>30</td>
<td>57</td>
</tr>
<tr>
<td>A-AyGaMoCo(0.6:5:3)</td>
<td>3.31</td>
<td>2.97</td>
<td>4.94</td>
<td>30</td>
<td>48</td>
</tr>
<tr>
<td>A-AyGaMoCo(0.6:5:4)</td>
<td>4.19</td>
<td>3.05</td>
<td>4.94</td>
<td>36</td>
<td>(48)</td>
</tr>
<tr>
<td>A-AyGaMoCo(0.6:5:5)</td>
<td>4.96</td>
<td>3.05</td>
<td>4.94</td>
<td>36</td>
<td>(48)</td>
</tr>
<tr>
<td>B-AyGaMoCo(0.6:5:1)</td>
<td>1.17</td>
<td>2.86</td>
<td>4.80</td>
<td>23</td>
<td>48</td>
</tr>
<tr>
<td>B-AyGaMoCo(0.6:5:2)</td>
<td>2.30</td>
<td>2.80</td>
<td>4.90</td>
<td>25</td>
<td>54</td>
</tr>
<tr>
<td>B-AyGaMoCo(0.6:5:3)</td>
<td>3.07</td>
<td>2.80</td>
<td>5.12</td>
<td>24</td>
<td>48</td>
</tr>
<tr>
<td>B-AyGaMoCo(0.6:5:4)</td>
<td>4.21</td>
<td>3.20</td>
<td>5.08</td>
<td>25</td>
<td>52</td>
</tr>
<tr>
<td>B-AyGaMoCo(0.6:5:5)</td>
<td>4.80</td>
<td>3.28</td>
<td>5.12</td>
<td>24</td>
<td>48</td>
</tr>
</tbody>
</table>

---

a Analytical, see text.  
b Due to the presence of Co₃O₄, the value is less accurate.  
c Value not calculated due to the large amount of Co₂O₃ present.

---

3. Experimental Results

3.1. Reflectance Spectra. Reflectance spectra were recorded for all samples. As the pattern of their spectra is essentially the same only representative series are reported. Figures 1 and 2 show the reflectance spectra of the AγGaCo(0.6:x) and A-AγGaMoCo(O.6:5:x) series, respectively. Detailed analysis of the reflectance spectra of Co²⁺ in different environments and of the assignment of optical transitions was discussed elsewhere.² Inspection of Figures 1 and 2 leads to the following conclusions:

(a) The spectra of the AγGaCo and AγGaMoCo specimens are dominated by bands due to Co²⁺ ions in tetrahedral symmetry² and their general pattern is qualitatively similar to that of the spinel C₆Mg₆Al₂O₁₂ and of Co⁺⁺/γ-Al₂O₃.⁵

(b) With increasing cobalt content the intensity of the absorption band at 578 nm is lower in the series AγCoₓ (with x ≤ 3 atom % since CoOₓ is present at higher x) than in the AγGaCo(0.6:x) series, Figure 3, as well as for the AγGaCo(4:x) series.

Recalling that the band at 578 nm is the most intense band for Co²⁺₄₄, one is led to the conclusion that the amount of Co²⁺₄₄ is higher in Ga-containing specimens.

(c) Moreover, inspection of Figure 3 shows that the intensity of the band at 578 nm increases linearly with the Co content for AγGaCo(0.6:x) as well as for AγGaCo(4:x) and A-AγGaMoCo(0.6:5:x), while this is not the case for B-AγGaMoCo, for which the curves become concave toward the abscissa at high Co content. This can be attributed to the presence of a new type of Co²⁺ with different symmetry. Since the extinction coefficient for octahedral Co²⁺ ions is smaller than for tetrahedral ions, the new type of Co²⁺ absorption can be attributed to ions in octahedral symmetry, probably in a new phase (the x-ray section clarifies this point).

(d) It is now useful to analyze the intensity of the shoulder at 720 nm (Co₂O₃) vs. cobalt content (atom percent): (△) AγCoₓ; (○) AγGaCo(0.6:x); (■) B-AγGaMoCo(0.6:5:x); (●) AγGaCo(4:x); (□) A-AγGaMoCo(0.6:5:x).

The gallium-free AγCoₓ and AγMoCo(5:x) have a larger amount of Co₂O₃, as compared to the gallium-containing specimens.
specimens AyGaCo(0.63:2) and Ay-AyGaMoCo(0.66:2) (with $x = 1$–8). Note that the samples of series B-AyGaMoCo do not show the presence of CoO$_4$.

In conclusion, the reflectance spectra show that the presence of Ga$^{+4}$ ions hinders the formation of CoO$_4$; moreover it favors a more normal CoAl$_2$O$_4$ spinel, i.e., a larger amount of Co$^{2+}$ in tetrahedral sites, except for the B series in which formation of CoMoO$_4$ is favored (see x-ray section).

3.2. X-Ray Measurements. The phase identification via x-ray spectra for supported catalysts presents difficulties. However, the analysis of the results, as far as the presence of the surface spinel CoAl$_2$O$_4$ and/or CoO$_4$ is concerned, can be made along lines similar to those discussed elsewhere$^{26}$ by comparing the intensities of different reflections, and, in more detail, the intensity profiles. The x-ray findings parallel the spectroscopic results.

It is useful to examine in more detail the x-ray spectra of AyGaMoCo catalysts according the method of preparation.

Method A. The x-ray pattern of A-AyGaMoCo(0.66:2) and A-AyGaMoCo(4:5:2) catalysts indicates only the formation of the surface spinel CoAl$_2$O$_4$.

Method B. The formation of CoAl$_2$O$_4$ as a surface spinel is confirmed. Moreover, additional lines are visible for Co $\geq$ 2 atom % and their intensity increases with cobalt content. These lines are attributable to CoMoO$_4$ phases; the lines at $d$ spacing = 6.25, 3.50, 3.12, and 2.09 Å to the green CoMoO$_4$ and the line at $d$ spacing = 3.36 Å to the violet CoMoO$_4$, called the "B" phase by Richet$^{25}$ and $\delta$ phase by Sleight and Chamberland.$^{11}$ In particular, in the specimens with 0.6 atom % Ga both phases are present, while in specimens with 4 atom % Ga only the violet phase is present. Since this phase can be transformed into the green phase by grinding,$^{10,12}$ we ground B-AyGaMoCo(4:5:2) for several hours. As a consequence, the line characteristic of the violet phase disappeared and the lines characteristic of the green phase appeared. Both CoMoO$_4$ phases were prepared to check the x-ray spectra.

Sulfided Catalysts. Sulfided A-AyGaCo(0.6:2) and A-AyGaMoCo(0.6:5:2) show spinel phase lines sharper than the corresponding oxidized specimens, without any additional lines. By way of contrast, the sulfided B-AyGaMoCo for Co $>$ 2 show, besides sharper spinel phase lines, new lines at $d$ = 5.67, 2.98, 1.91, and 1.75 Å, attributable to Co$_2$S$_2$ (ASTM index). No lines attributable to MoS$_2$ appear; note that the lines of CoMoO$_4$ disappeared.

3.3. Magnetic Measurements. Tables I and II report the Curie constant $C$, the Weiss temperature $\theta$, and the magnetic moment $\mu$ calculated from the Curie-Weiss law $\chi = C/(T - \theta)$, where $\chi$ is the magnetic susceptibility per mole of cobalt (actual analytical content) after correction for the diamagnetic contribution of all components.

Figure 5 reports the variation of $C$ with cobalt content for some representative specimens and includes the data for AyCox and AyMoCo(5:2) for comparison.

From inspection of Figure 5 and Tables I and II it appears that the $C$ values follow different trends according to whether gallium ions are present or absent and to the preparation method.

In principle, the $C$ value is dependent on two distinct facts: (a) the presence of different phases, such as CoO$_4$ ($C \approx 1$),$^{13}$ CoAl$_2$O$_4$ ($C_{calt} = 0.60^{14}$ and $C_{calt} = 2.67$) or CoMoO$_4$ ($C = 3.54$); (b) the distribution of Co$^{2+}$ ions among A (tetrahedral) and B (octahedral) sites of the surface spinel CoAl$_2$O$_4$; a higher $C$ value corresponds to a higher Co$^{2+}_{oct}$/Co$^{2+}_{tet}$ ratio.

It is then appropriate to examine the magnetic results in order to establish whether the variations in $C$ correspond to the presence of these different phases or to a variation of the Co$^{2+} _{oct}$/Co$^{2+} _{tet}$ ratio in the surface spinel CoAl$_2$O$_4$.

(a) Since only CoAl$_2$O$_4$ is present at low cobalt content (Co $\leq$ 2, above), the $C$ values depend on Co$^{2+}$ ion distribution among A and B sites of the surface spinel CoAl$_2$O$_4$. Thus, comparison of AyCox with AyGaCo(0.6:2) (curve a’ and a’ Figure 5) indicates that the presence of Ga$^{+4}$ ions favors a more normal spinel; in fact, a smaller $C$ value corresponds to a higher Co$^{2+} _{tet}$ content. (b) At higher cobalt content (Co $\geq$ 2), the $C$ values strongly decrease for the sample of series AyCox (curve a) and AyMoCo(5:2) (curve b), due to the presence of the phase Co$_3$O$_4$, while it remains constant (curve a’ and b’) for samples of series AyGaCo(0.6:2) and A-AyGaMoCo(0.6:5:2). In this last case, they reflect the distribution of Co$^{2+}$ ions among A and B sites of spinel CoAl$_2$O$_4$, this being the major Co species present. (c) As for the specimens of series B-AyGaMoCo, the magnetic data (curve b”, Figure 5) shows an increase of $C$ up to $C = 2$. This is due to the building up of the CoMoO$_4$ phase in which Co$^{2+}$ ions occupy octahedral sites.

Estimate of the Degree of Inversion from Magnetic Data. An estimate of the cobalt ions distributed in octahedral and tetrahedral sites can be made for those samples in which the Co$^{2+}$ ions can be assumed to be present as the surface spinel CoAl$_2$O$_4$; for this purpose we will then neglect the samples containing amounts of CoO$_4$.

For the B-AyGaMoCo catalyst, containing the surface spinel CoAl$_2$O$_4$ and CoMoO$_4$, it is also possible to estimate the amount of tetrahedral and octahedral cobalt ions. However, we can assume that the tetrahedral cobalt is present completely as the surface spinel CoAl$_2$O$_4$, for the octahedral cobalt we cannot determine how much is present as CoAl$_2$O$_4$ and how much as CoMoO$_4$. From the experimental value of the Curie constant, $C_{calt}$, and taking into account these restrictions, it is possible to use the law of additivity for computing the fraction of Co$^{2+}$ ions in octahedral and tetrahedral sites.$^{5,8}$ The results obtained using $C_{calt} = 3.56^{14}$ and $C_{calt} = 2.57^{15}$ are reported in Tables I and II.

It is necessary to emphasize that the percent values given for $\text{Co}^{2+}_{\text{oct}}$ are only estimates, because of the choice of $\text{C}_{\text{oct}}$ and $\text{C}_{\text{tet}}$, but the relative effect should be real.

Discussion

The presence and amounts of $\text{Co}_3\text{O}_4$, $\text{CoMoO}_4$, and $\text{CoAl}_2\text{O}_4$ are not determined solely by the chemical composition but also by the method of preparation, namely, A and B. Furthermore, the results show that the surface spinel $\text{CoAl}_2\text{O}_4$ is always present and its cation distribution is affected by the presence of foreign ions (i.e., in this case, $\text{Ga}^{3+}$ ions). Moreover, segregation of the $\text{Co}_3\text{O}_4$ phase depends on several factors, such as the cobalt content, the method of preparation, and the presence of $\text{Ga}^{3+}$ ions.

We discuss in order the following topics: (1) the effect of gallium ions on the cobalt and molybdenum; (2) the state of the surface according the method of preparation; (3) the sulfided specimens.

Influence of Gallium Ions on Cobalt and Molybdenum

In principle, in a spinel the $\text{M}^{2+}$ ions can occupy tetrahedral (A) and/or octahedral (B) sites and the relative $\text{M}^{2+}_{\text{tet}}/\text{M}^{2+}_{\text{oct}}$ ratio depends on several parameters. In addition, recent studies on the structural and magnetic properties of $\text{Ni}^{2+}$ ions supported on alumina showed that a small addition of $\text{Zn}^{2+}$, $\text{Ga}^{3+}$, or $\text{Ge}^{4+}$, all having a preference for tetrahedral sites, shifts the cation distribution in the surface spinel $\text{NiAl}_2\text{O}_4$ toward a more normal one.

The observed behavior was explained by invoking the polarization of anions toward tetrahedral sites. With this in mind, one would expect that the addition of cobalt ions to gallium-containing alumina, $\text{A}+\text{GaCo}$ specimens, leads to a more normal $\text{CoAl}_2\text{O}_4$ spinel. The experimental observation matches this picture fully. Especially in the specimens with high cobalt content, the experimental data (Figure 4) show that $\text{Co}_3\text{O}_4$ segregation is strongly decreased in the Ga-containing specimens ($\text{A}+\text{GaCo}$ as compared to $\text{A}+\text{Co}$). In order to rationalize this point, we recall that segregation of oxides in supported systems is mainly affected by two factors: (a) the stability of ions in the 2+ oxidation state; and (b) the diffusion pathways in the alumina lattice. It has been shown that for cations at tetrahedral sites the possible pathways in a spinel lattice always include a saddle, position of octahedral symmetry. Now, if the crystal field around the octahedral sites is decreased when $\text{Ga}^{3+}$ ions are added, one would expect a higher rate of diffusion of $\text{Co}^{2+}$ ions. As a consequence, a greater amount of spinel is formed while the segregation of $\text{Co}_3\text{O}_4$ decreases.

The presence of $\text{Ga}^{3+}$ may also influence the attachment of molybdate ions on the alumina surface. Molybdate monolayer formation has been extensively discussed by several authors. In our case it may be recalled that OH groups bonded to aluminum ions are about 100 times more basic than those bonded to gallium ions. Consequently, one would expect that molybdenum acid would react preferentially with aluminum octahedral OH giving molybdate ions attached to the surface. For a topotactic process, the molybdenum ions would occupy tetrahedral sites by extending the spinel structure immediately above the plane.

As compared to $\text{A}+\text{Mo}$, the $\text{A}+\text{GaMo}$ specimens will have a somewhat larger amount of $\text{Mo}^{6+}$ in tetrahedral sites and this fact may be relevant for catalytic reactions.

Influence of the Method of Preparation. Method A

This method consists of three successive impregnations and three calcinations according to the sequence Ga, Mo, and Co.

The first addition of Ga modifies the properties of the alumina surface. The second impregnation and calcination allows the Mo to react with the modified surface of the alumina giving a monolayer of molybdate ions attached to the alumina surface, and registry with the structure. In our case no separate phase of $\text{MoO}_3$ can be identified since the molybdenum content is fairly low. The cobalt, added with the third impregnation, now finds the structure of the external layers of alumina altered by the presence of $\text{Ga}^{3+}$ and molybdate ions.

It may be noted that the presence of $\text{Ga}^{3+}$ ions still allows the cobalt ions to react with alumina and favors a more normal spinel.

Finally, it should be emphasized that there is not a tendency to form the compound $\text{CoMoO}_4$, since all the molybdenum has already reacted with the alumina surface.

Method B. Since $\text{A}+\text{GaMoCo}$ catalysts were prepared by three impregnations but only two calcinations, the cobalt and molybdenum, react simultaneously (not in succession, as described for method A) with the alumina surface, and with each other. In fact, three reactions at 600 °C are able to occur simultaneously: (1) the reaction between Co- and Ga-containing alumina will give the cobalt aluminate, $\text{CoAl}_2\text{O}_4$, and $\text{Co}_3\text{O}_4$ at high Co content; (2) the reaction between molybdenum and alumina will form molybdate ions attached to the alumina surface; (3) the reaction between molybdenum and cobalt forms the $\text{CoMoO}_4$ phase.

The first reaction is dependent on the diffusion of cobalt ions into the external layers of alumina. The large surface area of alumina assists the process, by increasing the contact area between reagents.

The second reaction, considered as an acid–base reaction, should be dependent on the strength of the relative acids and bases involved and on the dispersion of molybdenum.

As far as the third reaction, the results obtained by Haber and Ziolikowski for the system $\text{Co}_3\text{O}_4$–$\text{MoO}_3$ at 500 °C clearly indicate that the $\text{CoMoO}_4$ formation is rapid and dependent on the diffusion of molybdenum ions into $\text{Co}_3\text{O}_4$ grains. Thus, the species expected to arise as a function of Co content when the three reactions occur simultaneously can be accounted for as follows.

At low cobalt content, only the first and the second reactions occur. Apparently, the third process ($\text{CoMoO}_4$ formation) is also very rapid at 600 °C although the strong interaction between cobalt and alumina tends to decompose the $\text{CoMoO}_4$ when this compound is heated with pure alumina.

Only at higher cobalt content does the formation of cobalt molybdate also occur, as a process taking place on the Co-rich external layers in competition with the diffusion of cobalt into the alumina. The experimental results match this picture fully. The x-ray data have shown that the formation of the $\text{CoMoO}_4$ phase starts to occur for a cobalt content ≥ 2 and its amount increases with increasing cobalt content. The reflectance spectra and magnetic measurements parallel the x-ray data showing that the cobalt ions, at higher cobalt content, go into octahedral sites as expected if $\text{CoMoO}_4$ is formed.

Sulfided Catalysts. As already reported, the effect of sulfiding is confined to the surface layers. The process can be pictured as an exchange of surface OH to SH with a concomitant reduction of $\text{Mo}^{6+}$ to lower oxidation states.

To throw light on the Co and Mo species which can be formed during the sulfurization, one has to take into account the species present in the oxide state.
According to literature data\(^2\),\(^3\),\(^21\),\(^35\),\(^36\) and to our results, the CoAl\(_2\)O\(_4\) cannot be sulfided, but only their surface OH\(^-\) can be transformed into SH\(^-\). However, if cobalt is present as separate phases (CoO\(_2\) or CoMoO\(_4\)) it will undergo the CoO\(_2\)S\(_2\) transformation.\(^2\),\(^3\),\(^25\),\(^28\),\(^30\) Concerning molybdenum, it is currently reported that the sulfiding process of the Mo monolayer leads to an "oxysulfido" species rather than to a sulfided one.\(^2\),\(^4\),\(^25\) This conclusion is based on the lack of observation of MoS\(_2\). However, it must be pointed out that the sulfiding of B-AgMoCo specimens, in which CoMoO\(_4\) is present, produces only CoS\(_2\), the conditions being suitable for the formation of CoO\(_2\)S\(_2\) and MoS\(_2\). Indeed the sulfiding of pure and silica supported (Co-Mo/SiO\(_2\)) CoMoO\(_4\) leads to the formation of cobalt and molybdenum sulfides.\(^2\),\(^3\) Therefore, it can be argued that the growth of MoS\(_2\) crystals on alumina supported specimens probably occurs in two dimensions, failing x-ray detection. Another possibility could be that the crystallites are three-dimensional, but very small. Indeed, XPS studies show that MoS\(_2\) is probably present.\(^2\),\(^7\),\(^30\)

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References and Notes


Analysis of Torsional Spectra of Molecules with Two Internal \(C_{3v}\) Rotors. 7

Far-Infrared and Low-Frequency Raman Spectra of the Gaseous Dimethylphosphine

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The Raman spectrum of gaseous dimethylphosphine at a resolution of 1 cm\(^{-1}\) has been recorded between 100 and 400 cm\(^{-1}\). The far-infrared spectrum has been recorded over the same frequency range with a resolution of 0.5 cm\(^{-1}\). Considerable torsional data are reported and used to characterize the torsional potential function based on a semirigid model. The average effective \(V_2\) was found to be 701 ± 3 cm\(^{-1}\) (2.01 kcal/mol). The cosi–cosi coupling term, \(V_{33}\), was found to be 240 ± 8 cm\(^{-1}\) (0.68 kcal/mol) and the sine–sine term, \(V_{33}\), has a value of -55 ± 2 cm\(^{-1}\) (-0.16 kcal/mol). These data are compared to the corresponding quantities obtained from microwave data. Comparisons are also given to similar quantities obtained for other molecules.

Introduction

The vibrational spectrum of dimethylphosphine has been previously reported.\(^1\),\(^2\) Beachell and Katlafsky\(^1\) reported the infrared spectrum of the gas and the Raman spectrum of the liquid and tentatively assigned the Raman lines at 236 and 318 cm\(^{-1}\) as the torsional motions. However, neither of these lines appeared in either the infrared or Raman spectra reported by Durig and Saunders\(^2\) and these authors concluded that these two bands arose from impurities in the earlier sample. Durig and Saunders\(^2\) reported bands in the far-infrared spectra of solid \((\text{CH}_{3})_2\text{PH}\) and \((\text{CD}_{3})_2\text{PH}\) at 180 and 140 cm\(^{-1}\), respectively, and assigned these bands to the \(\text{CH}_3\) and \(\text{CD}_3\) torsional modes. It was assumed that the frequencies for both torsional modes were essentially degenerate for both molecules and the threefold periodic barriers were calculated to be 2.14 and 2.30 kcal/mol for the \(\text{CH}_3\) and \(\text{CD}_3\) torsors, respectively. The barrier calculation was necessarily simplified because a proper characterization of the potential function for two interacting threefold torsors requires considerable experimental data.\(^2\) The availability of a

\(^{†}\) For part VI, see J. Chem. Phys., in press.