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Plasma-assisted atomic layer deposition of Ta$_2$O$_5$ from alkylamide precursor and remote O$_2$ plasma

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Tantalum oxide (Ta$_2$O$_5$) films were synthesized by plasma-assisted atomic layer deposition from pentakis(dimethylamino)tantalum (Ta[N(CH$_3$)$_2$]$_5$), precursor and remote O$_2$ plasma as oxidation source. Film growth was monitored in situ by spectroscopic ellipsometry, and film properties were investigated for deposition temperatures between 100 and 225 °C. Saturated precursor dosing conditions and plasma exposure times were identified and growth rates ranging from 0.8 Å/cycle at 225 °C to 0.87 Å/cycle at 100 °C were obtained. The deposited films were found to be stoichiometric (Ta:O=2:5). Moreover, no N incorporation was detected, and the C content was below the detection limit of the Rutherford backscattering measurement (<2 at. %) for all films studied. The mass density of the films, ranging from 7.8 g cm$^{-3}$ at 100 °C to 8.1 g cm$^{-3}$ at 225 °C, was found to be close to the bulk Ta$_2$O$_5$ density. The deviation could partly be accounted for by the amount of H detected with elastic recoil detection analysis, varying from 2 at. % at 225 °C to 4.6 at. % at 100 °C. X-ray diffraction revealed that all films were amorphous, independent of deposition temperature. The reaction mechanisms, in particular, during the plasma step, were investigated by using quadrupole mass spectrometry and optical emission spectroscopy. During the plasma step, combustion products such as CO, CO$_2$, and H$_2$O were detected. This indicates that combustion-like processes occur, in which the alkylamide N(CH$_3$)$_2$ ligands are oxidized by the O radicals generated in the plasma. Additionally, the presence of excited C≡N molecules in the plasma was observed in the plasma emission. © 2008 American Vacuum Society. [DOI: 10.1116/1.2905250]

I. INTRODUCTION

Tantalum oxide (Ta$_2$O$_5$) is an extensively studied material because of its interesting optical, chemical, and electrical properties. With a relatively high refractive index ($n \sim 2.2$) and a wide optical band gap (~4.35 eV), Ta$_2$O$_5$ is of interest for optical applications such as reflective coatings on mirrors and optical waveguides. Ta$_2$O$_5$ is also a very attractive material for use as a corrosion resistant coating due to its resistance to etching in various solutions. Due to its high dielectric constant ($k \sim 25$, for amorphous films), low leakage current ($<10^{-6}$ A/cm$^2$ at $\approx 3$ MV/cm), and voltage linearity, Ta$_2$O$_5$ has been employed as gate dielectric in metal-oxide-semiconductor field effect transistors, storage dielectric in dynamic random access memories (DRAMs), and insulating layer in thin film electroluminescent devices. Ta$_2$O$_5$ has proven to be especially suited as a dielectric in stacked metal-insulator-metal (MIM) capacitor structures that can be found in applications integrated in the back-end-of-line (BEOL), such as embedded DRAM and rf decoupling capacitors in radio frequency integrated circuits. A common method for the deposition of Ta$_2$O$_5$ is metal-organic chemical vapor deposition (MOCVD) by using Ta(OC$_2$H$_5$)$_5$ [(pentaoxo)ytantalum (PET)] precursor and O$_2$ gas at temperatures around 400 °C. The deposited films, however, relatively contain large amounts of C and H impurities and need to be annealed at temperatures of >600 °C to obtain the appropriate film properties. For low-thermal budget applications, such as MIM capacitors embedded in the BEOL and corrosion resistant coatings on temperature sensitive materials, there is a desire to minimize the heat load during Ta$_2$O$_5$ processing, making the current annealing step less compatible. Furthermore, the importance of the deposition of Ta$_2$O$_5$ on demanding three-dimensional (3D) structures has increased. This is, for example, due to the miniaturization of devices which requires higher capacitance densities and, consequently, the deposition of MIM capacitor stacks in trenches and pores. Another example is the surface...
finish of complex 3D objects by corrosion resistant Ta₂O₅. With MOCVD, it will become increasingly more challenging to grow thin, dense, and conformal Ta₂O₅ films at a low temperature on these structures once the aspect ratios employed increase. Therefore, alternative process techniques for the deposition of Ta₂O₅ are actively pursued.

Atomic layer deposition (ALD) has gained increasing interest for the growth of dielectric materials and is considered to be an attractive alternative technique to MOCVD for the growth of thin Ta₂O₅ films. In ALD, film growth is established through self-limiting surface reactions between alternately supplied gaseous precursors. The cyclic nature of ALD allows for excellent thickness control by the deposition of a single (sub)monolayer of film per cycle. Furthermore, the self-limiting surface reactions facilitate the growth of conformal and uniform films over large substrate areas with a challenging 3D topography. The possibility to deposit at relatively low temperatures is another major advantage of ALD.

By using H₂O as oxidant source, Ta₂O₅ has been deposited by the thermal ALD process from various Ta sources such as PET, and the metal halides TaCl₅ (tantalumpentachloride), TaF₅ (tantalumpentafluoride), and TaI₅ (tantalumpentaiodide). Deposition temperatures typically ranged from 200 to 400 °C. Alternatively, the growth of Ta₂O₅ from alternating exposures of PET and TaCl₅, i.e., without the use of an additional oxidant, has been demonstrated. By using a relatively high deposition temperature (>600 °C), it was also shown that films can be grown by using alternate exposures of TaI₅ and O₂ gas. Furthermore, the use of an O₂ plasma for oxidation (the so-called plasma-enhanced or plasma-assisted ALD) in combination with the PET precursor has been reported for both direct plasma and radical enhanced reactor configurations.

Recently, the use of the alkylamide-based precursor Ta[N(CH₃)₂]₅ [pentakis(dimethylamino)tantalum (PDMAT)] has attracted interest because of its high reactivity with surfaces during ALD, especially at low temperatures (<250 °C). Moreover, it has a relatively high vapor pressure, and the absence of Cl- and O-containing ligands results in noncorrosive reaction products which are more compatible with processing of metals and organic materials and are considered more hardware friendly. The film growth of Ta₂O₅ by plasma-assisted ALD using PDMAT and a remote O₂ plasma has recently been demonstrated by Maeng et al. A constant, relatively high growth rate of 1.2 Å/cycle was reported for deposition temperatures ranging from 150 to 250 °C. The material properties obtained with plasma-assisted ALD were found to be better compared to thermal ALD grown Ta₂O₅ from PDMAT by using H₂O as oxidant.

In this article, we report on the deposition of Ta₂O₅ by plasma-assisted ALD using PDMAT as Ta precursor and O₂ plasma as oxidant source. We have extended the analysis of the compositional, structural, and optical properties of Ta₂O₅ films to lower deposition temperatures, i.e., the 100–225 °C range. A major practical advantage of using a plasma as oxidant source is the possibility of keeping the cycle times short at low temperatures, since the reactivity is quickly removed when the plasma is switched off. In thermal ALD, the use of H₂O generally creates the necessity of applying a long purge period after the H₂O dose to maintain pure ALD film growth, i.e., without any chemical vapor deposition effect. In addition, experimental insight is presented into the reaction mechanism of plasma-assisted ALD of Ta₂O₅. While Ta₂O₅ material properties and the performance of devices employing this dielectric are extensively discussed in the literature, such experimental insight into the reaction mechanisms, in particular, during the O₂ plasma exposure, is lacking. Earlier, we have demonstrated that combustionlike reactions occur during the plasma-assisted ALD of Al₂O₃ from the alkyl metal-organic precursor Al(CH₃)₃ (trimethylaluminum) and O₂ plasma. The production of CO, CO₂, and H₂O was detected by mass spectrometry and optical emission spectroscopy (OES). Here, we present a similar experimental investigation during the O₂ plasma exposure of Ta₂O₅ deposition. Unlike the situation in Al₂O₃ deposition, where prior to the plasma exposure, the surface is covered with CH₃ groups resulting from the chemisorption of Al(CH₃)₃; in this case, N(CH₃)₂ surface groups are expected to cover the surface due to the binding of the Ta[N(CH₃)₂]₅ precursor to the surface. In the literature, it has been speculated that during the plasma exposure the formation of reaction products occurs, such as CO₂ and H₂O and, specifically for the alkylamide metal-organic precursors, also NO.

These products have been experimentally investigated in this work.

### II. EXPERIMENTAL

The experiments were carried out in a homebuilt remote plasma ALD reactor described in detail elsewhere. Briefly, it consists of a deposition chamber to which a precursor delivery system and remote inductively coupled plasma (ICP) source are attached. The chamber is pumped by a turbomolecular pump and has a base pressure of ~10⁻⁶ Torr. The typical processing conditions and experimental settings are summarized in Table I. The wall temperature was controlled at 85 °C. The solid metal-organic precur-

<table>
<thead>
<tr>
<th>Parameter</th>
<th>Standard setting</th>
<th>Range varied</th>
</tr>
</thead>
<tbody>
<tr>
<td>Ta[N(CH₃)₂]₅ dose time</td>
<td>3.0 s</td>
<td>(0.5–8 s)</td>
</tr>
<tr>
<td>Ar partial pressure</td>
<td>30 mTorr</td>
<td></td>
</tr>
<tr>
<td>Plasma exposure time</td>
<td>2.0 s</td>
<td>(0.5–5 s)</td>
</tr>
<tr>
<td>O₂ partial pressure</td>
<td>7.5 mTorr</td>
<td></td>
</tr>
<tr>
<td>Plasma power</td>
<td>100 W</td>
<td></td>
</tr>
<tr>
<td>Deposition temperature</td>
<td>225 °C</td>
<td>(100–225 °C)</td>
</tr>
<tr>
<td>Wall temperature</td>
<td>85 °C</td>
<td></td>
</tr>
<tr>
<td>Bubbler temperature</td>
<td>65 °C</td>
<td></td>
</tr>
</tbody>
</table>
resor Ta\text{[N(CH\text{3})\text{2}]}_{3}\) (PDMAT, Sigma-Aldrich, purity of >99.99%), stored in a stainless steel bubbler, was heated up to a temperature of 65 °C to reach an adequate vapor pressure. To prevent condensation, the PDMAT vapor was delivered to the chamber through heated supply lines (\(\approx 70 ^\circ\text{C}\)) by using a small Ar carrier gas flow controlled by a leak valve upstream. The Ar flow results in an operating pressure of \(\approx 30\) mTorr in the reactor and is also used to purge the chamber after the precursor dosing by using a divert line. In addition, the chamber was briefly pumped down (<\(10^{-4}\) Torr) after purging with Ar to remove any remaining gaseous precursor and reaction by-products before switching on the O\(_2\) flow through the plasma source. The ALD reaction was completed by exposure to an O\(_2\) plasma generated in the ICP source (100 W), operated at a pressure of 7.5 mTorr.

The films were deposited on \(4 \times 4\) cm\(^2\) native oxide covered monocrystalline Si (100) substrates (\(p\)-type, 10–30 \(\Omega\) cm) placed on a heated substrate holder (100–225 \(^\circ\text{C}\)). The substrates were exposed to an O\(_2\) plasma for 5 min prior to the deposition. During this plasma oxidation step of the substrate, the SiO\(_2\) film growth was monitored in situ by spectroscopic ellipsometry (SE) by using a J.A. Woollam, Inc., M2000U visible and infrared extended ellipsometer (0.75–5.0 eV), mounted under a 68° angle relative to the substrate normal. During Ta\(_2\)O\(_5\) processing, SE thickness measurements of the grown Ta\(_2\)O\(_5\) film were taken after completion of a certain number of ALD cycles. To investigate the optical properties of the deposited Ta\(_2\)O\(_5\) films near its band gap, an \(ex\ situ\) characterization was also performed by using a J.A. Woollam, Inc., M2000U visible and UV extended ellipsometer (1.2–6.5 eV), under an angle of 75°. A Tauc–Lorentz model was applied to fit the data by using the Woollam WAVE software.

The film composition was determined by Rutherford backscattering spectrometry (RBS), using a 2 MeV \(^4\text{He}^+\) beam under an 80° angle of incidence for determining the Ta, O, N, and C areal densities of the films. By using the same beam settings, elastic recoil detection (ERD) measurements were also performed to determine the H areal density. The mass density was calculated from the areal density by RBS, incorporating the density measurements were carried out by repetitively acquiring full spectra (350–870 nm) with an integration time of 100 ms. The spectral data of each spectrum were stored in separate files. For data reduction, the option in the software for selecting a single wavelength in the spectrum and acquiring the signal height as a function of time was used to monitor excited plasma species as a function of time during the plasma exposure step.

III. FILM GROWTH AND MATERIAL PROPERTIES

For optimizing the process, the precursor dose time and plasma exposure time were varied at a deposition temperature of 225 °C. Ta\(_2\)O\(_5\) film growth was monitored by \(in\ situ\) SE, directly yielding the growth rate from the film thickness plotted versus the number of cycles during a single deposition run. Under all deposition conditions, the film growth was observed to linearly proceed with the number of cycles. In Fig. 1(a), the growth rate can be seen to vary with the PDMAT dose time, reaching a maximum of 0.81 ± 0.02 Å/cycle for dose times of 3 s or more. A similar trend can be observed when the plasma exposure time is varied while keeping the PDMAT dose at 3 s. Saturated film growth was found to occur when the plasma is applied longer than 1 s [Fig. 1(b)]. The presence of a saturated growth regime also proved that the combination of Ar purge, the intermitting pump down of the reactor, and the switching of the gases (Ar to O\(_2\)) is adequate to avoid the occurrence of residual CVD processes.

The saturated growth rate of \(\approx 0.8\) Å/cycle obtained in our investigation considerably differs from the \(\approx 1.2\) Å/cycle reported earlier by Maeng et al. for the use of PDMAT and a remote O\(_2\) plasma in the temperature range of 150–250 °C. The exact reason for this deviation remains unclear; the main difference is that they used a higher plasma power (300–500 W) compared to the 100 W plasma power used in the current experiments. Variations in growth rates are, however, not uncommon in the ALD processing of metal oxides and tend to depend also on the oxidant source, the Ta source, and the reactor configuration used in the experi-
For instance, a more moderate growth rate of ≈1.0 Å/cycle has been reported for plasma-assisted ALD of Ta2O5 involving the use of the PET precursor and a direct O2 plasma. At the same time, a considerable higher growth rate, even up to 2.0 Å/cycle, has been reported for radical enhanced ALD using PET and an O radical source by Niskanen et al.22

The Ta2O5 film properties are presented in Table II for depositions at different temperatures under saturated growth conditions, i.e., 3 s precursor dosing and 2 s plasma exposure. Stoichiometric films (Ta:O=2:5) were obtained at all deposition temperatures, as revealed by RBS. Possible C and N impurities are below the RBS detection limits of 2 and 0.5 at. %, respectively. ERD revealed a H content ranging from 4.6 at. % at 100 °C to 2 at. % at 225 °C. Although not directly measured here, it is conceivable that the H is incorporated in the form of –OH, which is typically reported for oxide films grown by ALD and CVD processes at low temperatures. The mass density of the films is calculated from the RBS elemental composition and the SE thickness and is found to increase with the deposition temperature from 7.8 g cm⁻³ at 100 °C to 8.1 g cm⁻³ at 225 °C. The latter value is close to the reported Ta2O5 bulk value of 8.24 g cm⁻³.34 The growth rate shows a decrease from 0.87 ± 0.02 Å/cycle at 100 °C to 0.81 ± 0.02 Å/cycle at 225 °C. While the growth rate decreases with temperature, the number of Ta atoms deposited per cycle roughly remains the same (∼0.18 × 10¹⁵ at. cm⁻²/cycle) at all temperatures (cf. Table II). The variation of growth rate with temperature can, therefore, be largely explained by the difference in mass density of the deposited films. This is remarkably different from other plasma-assisted ALD processes, such as Al2O3, where the number of deposited Al atoms per cycle decreases with increasing deposition temperature.35

In Fig. 2, x-ray diffraction (XRD) spectra are shown for the Ta2O5 films reported in Table II. Besides the large c-Si substrate peak at 2θ=69° for Si (400), a broad feature between 20° and 40° can be observed for all samples. The absence of sharp peaks in the XRD spectrum indicates that the Ta2O5 films are primarily amorphous when deposited in the temperature range of 100–225 °C. This kind of microstructure is typically found in XRD spectra of ALD Ta2O5 films in this temperature range for both metal-organic and metal-halide based Ta precursors used in combination with H2O.14,36 For deposition temperatures above 300 °C, the hexagonal β-Ta2O5 phase is typically found in films deposited from metal-halide Ta precursors (i.e., TaCl5, TaF5, and Ta3) in combination with H2O.14–16 At increased deposition temperatures or upon annealing, the Ta2O5 films tend to be-

**Table II.** Film properties as determined by Rutherford backscattering spectrometry, elastic recoil detection, and spectroscopic ellipsometry for Ta2O5 films grown by using 700 ALD cycles at different deposition temperatures. The optical band gap $E_g$ has been determined from the spectroscopic ellipsometry data by using the Tauc formalism.

<table>
<thead>
<tr>
<th>Properties</th>
<th>100 °C</th>
<th>150 °C</th>
<th>225 °C</th>
</tr>
</thead>
<tbody>
<tr>
<td>Thickness (Å)</td>
<td>607 ± 2</td>
<td>559 ± 2</td>
<td>568 ± 2</td>
</tr>
<tr>
<td>Growth rate (Å/cycle)</td>
<td>0.87 ± 0.02</td>
<td>0.80 ± 0.02</td>
<td>0.81 ± 0.02</td>
</tr>
<tr>
<td>Ta (10¹⁵ at. cm⁻²/cycle)</td>
<td>0.185 ± 0.007</td>
<td>0.173 ± 0.007</td>
<td>0.180 ± 0.007</td>
</tr>
<tr>
<td>O (10¹⁵ at. cm⁻²/cycle)</td>
<td>0.462 ± 0.007</td>
<td>0.434 ± 0.007</td>
<td>0.449 ± 0.007</td>
</tr>
<tr>
<td>H (10¹⁵ at. cm⁻²/cycle)</td>
<td>0.031 ± 0.007</td>
<td>Not measured</td>
<td>0.014 ± 0.007</td>
</tr>
<tr>
<td>Ta:O</td>
<td>2:5 ± 0.1</td>
<td>2:5 ± 0.1</td>
<td>2:5 ± 0.1</td>
</tr>
<tr>
<td>H (at. %)</td>
<td>4.6 ± 1</td>
<td>Not measured</td>
<td>2.0 ± 1</td>
</tr>
<tr>
<td>Mass density (g cm⁻³)</td>
<td>7.8 ± 0.05</td>
<td>8.0 ± 0.05</td>
<td>8.1 ± 0.05</td>
</tr>
<tr>
<td>$E_g$ (eV)</td>
<td>4.31 ± 0.05</td>
<td>4.29 ± 0.05</td>
<td>4.26 ± 0.05</td>
</tr>
<tr>
<td>n (at 2.0 eV)</td>
<td>2.20 ± 0.03</td>
<td>2.20 ± 0.03</td>
<td>2.23 ± 0.03</td>
</tr>
</tbody>
</table>
come more crystalline and changing toward the orthorhombic $\beta$-Ta$_2$O$_5$ phase with the undesirable effect of causing a higher leakage current. At all deposition temperatures and even after annealing, the films are reported to contain a large fraction of amorphous material.

The refractive index ($n$) and extinction coefficient ($k$) of the deposited Ta$_2$O$_5$ were determined from the fit of the ellipsometric data of the films described in Table II. In Fig. 3, the $n$ and $k$ of the Ta$_2$O$_5$ film deposited by plasma-assisted ALD at 225 °C are shown over a range of 0.75–6.5 eV (1700–193 nm). The data have been generated by using the Tauc–Lorentz parametrization of the combined data as obtained by the infrared (down to 0.75 eV) and UV (up to 6.5 eV) extended ellipsometers. The Tauc–Lorentz fit parameters at 225 °C were found to be $A=330.2$, $E_n=4.950$ eV, $C=1.769$ eV, $E_g=4.216$ eV, and $E_{\text{inf}}=2.312$ eV, and these values were found to be very similar for all films. At a photon energy of 2 eV ($\sim$633 nm), the refractive index is found to range from $n=2.20$ at 100 and 150 °C to $n=2.23$ at 225 °C. The refractive index is close to the highest reported values ($n=1.9–2.2$) for Ta$_2$O$_5$ ALD using various Ta precursors and H$_2$O as oxidant in the same temperature range. The high refractive index can be correlated to the relatively high density of the Ta$_2$O$_5$ material obtained in this work. From the SE data, the Tauc band gap was determined. Tauc band gap values ranged from $E_g=4.31$ eV at 100 °C to $E_g=4.24$ eV at 225 °C. These values are close to typical values reported for thermal ALD Ta$_2$O$_5$ films.

IV. REACTION MECHANISM STUDIES

To investigate the reaction mechanisms governing Ta$_2$O$_5$ film growth by plasma-assisted ALD, QMS and OES were employed. In particular, the reactions occurring during the O$_2$ plasma exposure are of interest since a clear difference with the thermal ALD process exists in the form of the chemical reactivity of the oxidant source. The deposition temperature during the QMS and OES experiments is set by the wall temperature (85 °C) of the deposition chamber. Saturated dosing conditions were used (3 s PDMAT dose and 5 s plasma exposure), unless otherwise stated.

QMS data of the mass-to-charge ($m/z$) ratios in the range of $m/z=15–45$ were obtained by using the experimental procedure described in Sec. II. In Fig. 4(a), time dependent scan over the two ALD cycles is shown. The periods during which the Ar flow and the O$_2$ flow are active have been indicated, as well as the precursor dosing time and the plasma exposure time.

During the precursor dose, no response was observed at any of the $m/z$ signals shown in Fig. 4. As a matter of fact, no response to the PDMAT precursor introduction was seen.
at any of the $m/z$ signals in the range of $m/z=15$–45. This is remarkable since this would imply that there is no production of HN(CH$_3$)$_2$ which would suggest that another chemisorption mechanism other than the commonly presumed ligand exchange reaction would be occurring. A lack of detection sensitivity of the QMS for the measurements carried out could be another explanation for the absence of a signal due to HN(CH$_3$)$_2$. However, in a similar investigation of plasma-assisted deposition of TaN from PDMAT and H$_2$ plasma, where the same mass spectrometer was used and under similar experimental conditions, a clear response was observed during the precursor dosing at $m/z=28$, 44, and 45. Under these conditions the response closely matched the cracking pattern of HN(CH$_3$)$_2$ $[m/z=44$ (100%), 45 (64%), and 28 (18%)]. In the case of plasma-assisted ALD of TaN, the chemisorption of PDMAT most likely involves binding to NH$_2$ surface groups by splitting off volatile HN(CH$_3$)$_2$ reaction products. A lack of sensitivity of the mass spectrometer in the Ta$_2$O$_5$ experiments is, therefore, not likely. In addition, we would like to note that the chemisorption of metal alkylamides has not been as extensively studied as for metal-alkyl or metal-halide precursors. On the basis of quartz-crystal microbalance measurements, Hausmann et al. speculated about the chemisorption of PDMAT on a hydroxylated surface during thermal ALD of Ta$_2$O$_5$; however, the limited experimental data prohibited the exact identification of the surface reactions. For plasma-assisted ALD processes of metal oxides from metal alkylamides, even less literature data have been reported and even no information is available about the surface composition of amorphous Ta$_2$O$_5$ after the oxidation step. This is contrary to the Al$_2$O$_3$ case, for which evidence has been presented that –OH groups are the main reactive surface sites both in thermal ALD with H$_2$O as oxidant and plasma-assisted ALD using an O$_2$ plasma. Thus, the exact precursor chemisorption mechanism occurring during the deposition of Ta$_2$O$_5$ remains an open question.

During the O$_2$ plasma step of the plasma-assisted ALD process of Ta$_2$O$_5$, clear peaks can be observed at $m/z=18$ and 44. These signals indicate the formation of H$_2$O and CO$_2$, respectively. Their formation can be attributed to combustionlike reactions during the plasma exposure induced by O radicals. Also, a clear peak can be distinguished at $m/z=28$, which is associated with the presence of CO that can be formed either via incomplete combustion reactions or through the electron impact dissociation of CO$_2$ in the plasma. Furthermore, from the dip in the $m/z=32$ (O$_2$) signal, the consumption of oxygen due to the oxidation of the Ta surface layer and simultaneous formation of H$_2$O, CO$_2$, and possibly CO can be deduced.

From the peak intensities of the signals shown in Fig. 4 and the other $m/z$ ratios measured in the range of $m/z=15$–45, the gas composition during the plasma exposure (at $t\sim 200$ ms after ignition of the plasma) was determined. The result is shown in Fig. 5 as a bar graph of the signals that increased during the plasma exposure compared to the QMS background. For some of the signals, the most likely contributing gas species have been indicated. Besides the increase at $m/z=28$ due to ionization and dissociative ionization of CO and CO$_2$ in the mass spectrometer, respectively, a clear increase can also be observed at $m/z=27$, 29, and 30. On the basis of the cracking patterns, we suggest that the main candidates contributing to the signals are HCN$^+$ ($m/z=27$), N(CH$_3$)$_2^+$ ($m/z=29$), and NO$^+$ ($m/z=30$), respectively. From the gas composition, we can speculate that the –N(CH$_3$)$_2$ surface groups after PDMAT chemisorption appear to be not completely combusted by the O radicals. Possibly, a combination of different surface processes simultaneously occurs. The latter was also observed in the plasma-assisted ALD process of Al$_2$O$_3$ from Al(CH$_3$)$_3$. In this process, both combustionlike reaction products (CO, CO$_2$, and H$_2$O) as well as hydrocarbon species (CH$_4$, C$_2$H$_6$, etc.) were observed during the O$_2$ plasma exposure. One reaction mechanism for hydrocarbon generation was a secondary reaction pathway in which the produced H$_2$O reacted with the –CH$_3$ surface groups through ligand exchange to form CH$_4$ and –OH groups.

OES was used to investigate the excited species in the gas phase during the plasma exposure. The light emissions which are observed during the first 200 ms of the plasma exposure in the plasma-assisted ALD processing of Ta$_2$O$_5$ and during a regular O$_2$ plasma in continuous operation are shown in Figs. 6(a) and 6(b), respectively. A clear difference can be observed between both spectra. The difference is related to the presence of reaction products in the plasma during the oxidation step of plasma-assisted ALD of Ta$_2$O$_5$. From an identification of the spectral lines, the broadband emission in Fig. 6(a) was found to be predominantly originating from excited CO$^+$ molecules. The bands from the Angström (B $^1\Sigma^+-A$ $^1\Pi$), Herzberg (C $^1\Sigma^+-A$ $^1\Pi$), and thrid positive ($b$ $^3\Sigma^+-a$ $^3\Pi$) systems of CO$^+$ as well as the Comet tail (A $^3\Pi$–X $^2\Sigma^+$) and Balder–Johnson (B $^3\Sigma^+-A$ $^3\Pi$) systems of the CO$^{++}$ ion were identified, as indicated in Fig. 6(a). Several atomic oxygen lines (777 and 845 nm) could also still be clearly identified in the spectrum during the Ta$_2$O$_5$ ALD process.

For comparison, the spectrum recorded during the plasma-assisted ALD of Al$_2$O$_3$ is shown in Fig. 6(c). The
spectra for Ta$_2$O$_5$ and Al$_2$O$_3$ show similarities, but a clear difference is, besides the higher broadband background, the presence of peaks at around 388 and 358 nm. These peaks have been identified to originate from the Violet system $B^1\Sigma_-^+ - X^1\Sigma_+^-$ of C$_2$N*, and they are clearly related to the presence of alkylamine ligands in PDMAT. The emission of excited nitric oxide $\text{NO}^*$ could not be observed in the spectra, although the formation of NO was indicated by the QMS measurements. The reason for this is that NO* mostly radiates in the UV region $\text{system: } 195.61–345.85 \text{ nm}$, which is below the UV cutoff of the optical emission window.

Since the reaction products are continuously pumped away, their characteristic emission can also be used to gain information on the progress of the surface reactions, thus serving as an indicator for the completion of the plasma step. Furthermore, since the number of reaction products is limited to the number of surface sites that underwent reactions, the OES measurements can also be used to investigate the actual film growth by varying one of the growth parameters such as precursor dose. From the emission spectra shown in Fig. 6, the wavelengths of several peaks related to the reaction products were selected and monitored as a function of time during the plasma exposure. In Fig. 7, the time dependent signals at 777 nm ($\text{O}^*$), 656 nm ($\text{H}^*$), 519 nm ($\text{CO}^*$), and 388 nm ($\text{CN}^*$) are given for different precursor dose times. Figure 7(a) shows that when no PDMAT dose ($t=0$ s) preceded, the plasma exposure signal at 777 nm immediately shows a stable $\text{O}^*$ emission after plasma ignition. When the precursor has been introduced for a relatively brief period of time ($t=0.5$ s) prior to plasma exposure step and, consequently, surface reactions have taken place, the $\text{O}^*$ emission is observed to be slightly lower at the start of the plasma exposure. Subsequently, it reaches its steady state level again after a certain period. With the introduction of more precursor ($t=2$ and 4 s), this effect is becoming increasingly pronounced. The decrease in intensity of the $\text{O}^*$ radical emission at the start of the plasma exposure correlates with the sharp increase seen in the $\text{CO}^*$, $\text{CN}^*$, and $\text{H}^*$ emissions shown in Figs. 7(b)–7(d). The increase in emission is followed by a decrease to its steady state level after some time. The shape of this transient signal depends on the production and removal rate of the excited reaction products. For all species, the transient signal becomes more pronounced with increasing precursor dose which indicates that more reaction products have been generated at larger precursor doses up to a precursor dosing time of 4 s.

The completion of the plasma step can be determined from the decay of the transient emissions of $\text{H}^*$, $\text{CO}^*$, and $\text{CN}^*$. Typical 1/e decay times of $\sim0.5$ s are found for the
transient signal of all species, revealing that the plasma step is completed within ~2 s (i.e., several 1/e decay times). A similar result is obtained from the O* emission when estimating the time in which it reaches its steady state value. A saturation time of <2 s correlates well with the SE film growth data, as presented in Fig. 1. The total light emission by the reaction product species can be determined by integrating the time dependent signals displayed in Fig. 7. The result is shown in Fig. 8, where the integrated emission signals of CO* and CN* are shown as a function of precursor dose. Some initial differences at low precursor doses can be observed between the integrated CO* and CN* emission signals, but a clear saturation of the amount of emission can be observed for both CO* and CN* for precursor doses larger than 3–4 s. The saturation of both CN* and CO* emissions at higher precursor dose is induced by the fact that during the plasma exposure, only a limited amount of surface groups enters the gas phase due to the self-limiting adsorption reactions of the precursor. When compared to the growth rate measured by SE shown in Fig. 1, the OES results for CO* and CN* show a very similar trend, reaching a saturated value when a precursor dose of 3–4 s and longer is applied. The comparison between the SE film thickness and the QMS and OES gas phase measurements forms a check. It correlates the gas phase measurements, to which all surfaces on which film growth occurs contribute to the local measurement of film growth on the substrate by SE. This check indicates that, due to the uniform temperature throughout the reactor and the deposition of a fresh Ta2O5 layer prior to this experiment, the film growth occurs at the same rate and most likely in the same fashion on all reactor surfaces. The difference in emission between CN* and CO* at a low precursor dose (τ<3–4 s) could be related to the amount of surface groups that is present due to the chemisorption of the precursor at the surface. We can assume that the plasma produces the same number of O radicals in any condition and, therefore, the following explanation can be suggested. At low precursor dose (τ<3–4 s), the number of surface groups is relatively small compared to the number of O radicals, such that the C atoms in the surface groups are mainly converted into CO and/or CO2. With increasing precursor dose, however, the density of surface groups increases and the ratio between the O radicals and surface groups decreases. In this case also, CN starts being formed due to the lower amount of O radicals available, although the combustion and the formation of CO and CO2 might still be the dominant process.

V. CONCLUSIONS

The Ta2O5 film growth by plasma-assisted ALD using PDMAT precursor and an O2 plasma has been investigated. The process was proven to exhibit true ALD-like properties showing saturated film growth at a rate of 0.81 Å/cycle at 225 °C. The films were stoichiometric (Ta:O=2:5) and N and C impurities were below the detection limit of the RBS analysis. The mass density was high and close to the bulk value of crystalline Ta2O5 (>98%), whereas the as-deposited films were found to be predominantly amorphous. While the growth rate in terms of thickness increase per cycle was found to slightly vary with deposition temperature, increasing to 0.87 Å/cycle at 100 °C, the number of Ta atoms deposited per cycle was found to remain independent of temperature. The difference in growth rate could be explained by the decrease in mass density. The H content in the film was found to vary from 4.6 at. % at 100 °C to 2.0 at. % at 225 °C. The optical properties in terms of refractive index and extinction coefficient were determined in the range of 0.75–6.5 eV by using SE. From these data, the Tauc band gap values were determined to range from Eg=4.31 eV at 100 °C to Eg=4.24 eV at 225 °C.

The associated reaction mechanisms of the ALD process have been studied by using QMS and OES. During the precursor dose, remarkably, no H(N(CH3)2) and associated cracking products in the range of m/z=15–45 were detected. This might suggest a different chemisorption mechanism from simple ligand exchange. For example, an association reaction of the complete precursor could take place during the PD-MAT step. In such an association reaction, the precursor remains intact when it chemically adsorbs without the release of reaction products into the gas phase. However, further investigations are necessary to settle this issue. During the plasma exposure, the occurrence of combustionlike reactions was established through the detection of CO, CO2, and H2O. The Ta2O5 plasma-assisted ALD process from the alkylamide-based Ta[N(CH3)2]5 precursor, therefore, showed many similarities with the deposition of Al2O3 from the alkyl-based Al(CH3)3 precursor previously reported. Specifically for the Ta2O5 process, however, the production of NO was found by mass spectrometry, indicating the presence of alkylamide surface groups [-N(CH3)2] after chemisorption of PDMAT. Furthermore, the detection of signals due to HCN+, NCH2+, and C≡N* by mass spectrometry and OES suggests that during the plasma exposure, other processes also occur at the surface, in addition to combustion reactions. On the basis of the investigation on Ta2O5 presented here and the earlier work on Al2O3, it can be concluded that the occurrence of combustionlike reactions during the plasma exposure appears to be generic for plasma-
assisted ALD processes of metal oxides from metal-organic precursors when using an O$_2$ plasma for oxidation.

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