B-spline parametrization of the dielectric function applied to spectroscopic ellipsometry on amorphous carbon

Citation for published version (APA):

DOI:
10.1063/1.3257237

Document status and date:
Published: 01/01/2009

Document Version:
Publisher’s PDF, also known as Version of Record (includes final page, issue and volume numbers)

Please check the document version of this publication:
• A submitted manuscript is the version of the article upon submission and before peer-review. There can be important differences between the submitted version and the official published version of record. People interested in the research are advised to contact the author for the final version of the publication, or visit the DOI to the publisher’s website.
• The final author version and the galley proof are versions of the publication after peer review.
• The final published version features the final layout of the paper including the volume, issue and page numbers.

Link to publication

General rights
Copyright and moral rights for the publications made accessible in the public portal are retained by the authors and/or other copyright owners and it is a condition of accessing publications that users recognise and abide by the legal requirements associated with these rights.

• Users may download and print one copy of any publication from the public portal for the purpose of private study or research.
• You may not further distribute the material or use it for any profit-making activity or commercial gain
• You may freely distribute the URL identifying the publication in the public portal.

If the publication is distributed under the terms of Article 25fa of the Dutch Copyright Act, indicated by the “Taverne” license above, please follow below link for the End User Agreement:
www.tue.nl/taverne

Take down policy
If you believe that this document breaches copyright please contact us at:
openaccess@tue.nl
providing details and we will investigate your claim.
B-spline parametrization of the dielectric function applied to spectroscopic ellipsometry on amorphous carbon

J. W. Weber, T. A. R. Hansen, a) M. C. M. van de Sanden, and R. Engeln b)

Department of Applied Physics, Eindhoven University of Technology, P.O. Box 513, 5600 MB Eindhoven, The Netherlands

(Received 14 July 2009; accepted 27 September 2009; published online 16 December 2009)

The remote plasma deposition of hydrogenated amorphous carbon (a-C:H) thin films is investigated by in situ spectroscopic ellipsometry (SE). The dielectric function of the a-C:H film is in this paper parametrized by means of B-splines. In contrast with the commonly used Tauc–Lorentz oscillator, B-splines are a purely mathematical description of the dielectric function. We will show that the B-spline parametrization, which requires no prior knowledge about the film or its interaction with light, is a fast and simple-to-apply method that accurately determines thickness, surface roughness, and the dielectric constants of hydrogenated amorphous carbon thin films. Analysis of the deposition process provides us with information about the high deposition rate, the nucleation stage, and the homogeneity in depth of the deposited film. Finally, we show that the B-spline parametrization can serve as a stepping stone to physics-based models, such as the Tauc–Lorentz oscillator. © 2009 American Institute of Physics. [doi:10.1063/1.3257237]

I. INTRODUCTION

The chemical inertness, low to high nanohardness, and thermal and conductive properties of hydrogenated amorphous carbon (a-C:H) thin films allow them to be used in a wide variety of applications, ranging from the microchip industry to protective (e.g., optical windows and magnetic storage disks) and biomedical coatings. 1,2 Remote plasma deposition is used to deposit, among others, diamondlike carbon films with a nanohardness in excess of 13 GPa at high deposition rates (>10 nm/s), while maintaining good adhesion and chemical stability (see, e.g., Refs. 1, 3, and 4, and references therein).

In this paper, remote plasma deposition of hydrogenated amorphous carbon (a-C:H) thin films is investigated by in situ spectroscopic ellipsometry (SE). SE is a noninvasive optical diagnostic that can measure the change in polarization of light reflected on a thin film. 5–7 This change in polarization is determined by the ratio of the Fresnel reflection coefficients for both p- and s-polarized light, commonly expressed as ρ=r_p/r_s=tan Ψ e^{iΔ}. Determining the dielectric function (ε=ε_1+iε_2) from the measured SE data requires a (multilayered) model that describes the interaction of the incident light with the film. The dielectric spectrum of an a-C:H film (see Fig. 1) is characterized by the π-π* electronic transition around 4 eV and the σ-σ* electronic transition around 13 eV. 8–10 In SE studies of these films, each transition is commonly modeled by a Tauc–Lorentz (TL) oscillator. 8,9,11 However, the majority of standard spectroscopic ellipsometers are not capable of reaching 13 eV or above, necessitating the use of complementary diagnostics, e.g., electron energy loss spectroscopy. 12,13 In situ measurements with such complementary diagnostics during film growth are in most cases, such as ours, practically not feasible. Also the (in)homogeneity in depth of the carbon film, which is not always known, needs to be taken into account in the optical model. For all these reasons, it is therefore not always possible to apply a model based on two TL oscillators to the measured SE data.

Even when two TL oscillators could be used, it is not necessary to apply such a physics-based model if only the — evolution in — thickness and roughness (i.e., growth rate and nucleation) and an accurate parametrization of the dielectric function of the thin film are of interest to the experimentalist. The approach in this paper, therefore, is to obtain a purely mathematical and Kramers–Kronig consistent parametrization of the dielectric function by means of B-splines. 14 Such a parametrization of the deposited carbon layer requires no prior knowledge about the film properties or assumptions about the interaction of light with the film. The optical model for our films consists of a substrate, the

---

a)Electronic mail: t.a.r.hansen@tue.nl.
b)Electronic mail: r.engeln@tue.nl.
bulk of the film itself, and a roughness layer, whereby each layer is defined by its own thickness and dielectric constants. While the dielectric function of the bulk layer is represented by B-splines, the roughness is modeled by Bruggeman’s effective medium approximation (EMA) of 50% bulk material and 50% voids. This layer structure, whereby the bulk carbon material is parametrized by means of B-splines, is hereafter referred to as the B-spline model. An ideal fit between model and data would give a value close to 1 for the unbiased maximum likelihood estimator $\chi^2$. Large deviations from 1 could indicate an incorrect or incomplete model. However, it can also point to the accumulation of very small experimental errors in the obtained data. A model can quickly grow in complexity, while still reducing the overall $\chi^2$ due to an increase in the number of fitting parameters. Therefore, we verify the validity of the layer structure of the B-spline model for our carbon films by complementary experimental techniques, i.e., atomic force microscopy (AFM) and cross-sectional scanning electron microscopy (SEM) measurements. In addition to verifying the layer structure, we will also compare the B-spline parametrization — of an ex situ measurement — with a wavelength-by-wavelength fit.

In situ measurements, also analyzed by the B-spline model, are used to calculate the deposition rate and to investigate the nucleation of the film. Although other methods — e.g., single wavelength ellipsometry — are available to determine the film thickness during deposition, in situ SE and, in particular, the B-spline model have the advantage that the dielectric function can be accurately determined without the need for any assumptions about the film’s interaction with light. This will allow us to establish the (inhomogeneity in depth of the deposited layer. We will also show that the B-spline parametrization of the dielectric function can also serve as a stepping stone to a parametrization with a TL oscillator. Finally, the dielectric spectra of the a-C:H film in both vacuum and ambient air are tabulated in this paper by means of B-splines.

II. EXPERIMENTAL SETUP

A. Reactor

Hydrogenated amorphous carbon thin films are deposited on a 1 mm thick silicon wafer with a 1.6 nm native oxide layer. The plasma source (see Fig. 2) used for deposition is a cascaded arc consisting of a stack of four water cooled copper plates with a 4 mm central arc channel. A dc of 75 A runs from three tungsten cathodes, through the arc channel, to the anode plate at the end. This current creates an Ar plasma (100 sccs) under high pressure (540 mbar) that expands supersonically into a low pressure reactor (30 Pa). After formation of a shock zone, the plasma continues subsonically toward the substrate holder, located at around 55 cm from the exit of the arc. The substrate temperature is kept at a constant 250 °C throughout the 25–40 s deposition time, which is controlled by the opening and (automatic) closing of the shutter. A He backflow of 1 sccs is used for improved thermal contact between the substrate holder and the sample.

B. Spectroscopic Ellipsometry

The reactor is equipped with a load lock, a shutter to avoid direct exposure of the sample to the plasma jet, and access ports for SE with a fixed angle of incidence of 68°. During ex situ SE measurements the angle of incidence is 70°. The precursor, 15 sccs of acetylene (C2H2), is injected through a ring, which is located 5 cm from the exit of the plasma source. The dissociation of the precursor occurs in the expanding plasma jet through charge transfer and dissociative recombination with the Ar ions and electrons, respectively. This ion chemistry results in a high radical flux toward the surface, which causes a high deposition rate.

The $\chi^2$ measurement rate of the ellipsometer is set to 200 rev/measurement for the ex situ measurement, with the high accuracy mode enabled. This indicates that each measurement is the average of 200 revolutions of the compensator. To obtain the highest acquisition rate for the in situ measurements, 1 rev/measurement without the high accuracy mode is used. The actual number of rev/measurement, over which is averaged, depends on the processing power of the computer. Although this number is not included in the data file, it can be determined from the acquisition time between two datapoints. The analysis software is WVASE32 3.668 and CompleteEASE 3.55 and 4.06, from J. A. Woollam Co., Inc.
1. Layer structure

Each optical model, used for the analysis of SE data, has a layer structure whereby each layer is defined by its own thickness and dielectric constants. A three-tiered layer structure is used for the a-C:H thin films investigated in this paper. The first layer is a 1 mm thick silicon wafer with a 1.6 nm native oxide layer. The native oxide layer was measured prior to deposition, and both thicknesses are fixed in the layer model. This is the substrate on which amorphous carbon, i.e., the second layer, is deposited. The third and last layer is the roughness, which is modeled by Bruggeman’s EMA of 50% bulk material and 50% voids. The (in)homogeneity in depth of the carbon layer is discussed in Sec. III C. This fairly simple and straightforward layer structure is used for the B-spline model, in which the dielectric function of the carbon layer is parametrized by means of B-splines.

2. B-splines

Basis-splines, commonly abbreviated to B-splines, are a recursive set of polynomial splines

\[ B_i^k(x) = \begin{cases} 1, & t_i \leq x \leq t_{i+1} \\ 0 & \text{otherwise}, \end{cases} \]

where \( k \) is the degree of the B-spline and \( i \) is the index for the knots \( t_i \) that denote the position, on the x-axis, where the polynomial segments connect. The total spline curve \( S(x) \), representing the dielectric function of the film layer, is then given by

\[ S(x) = \sum_{i=1}^{n} c_i B_i^k(x), \]

with \( c_i \) the B-spline coefficients. \( (t_{i+2}, c_i) \) denote the locations of the control points of the B-spline curve. If there are \( n \) knots then there are \( n-k-1 \) control points, i.e., the last control point is located at \((t_{n-k+1}, c_{n-k-1})\). CompleteEASE uses cubic B-splines (\( k=3 \)). With \( k=3 \), each control point influences only the two previous and the following two polynomial segments. This is known as local support.

The number of coefficients used to accurately describe the dielectric function should be kept as low as possible, while still adhering to the shape of the function. Although additional coefficients can provide a better description, too large a number leads to an unrealistic result in which only the noise is better described. It also increases the possibility of correlation between the coefficients.

B-splines can be ensured to have a physical meaning because of the following two properties. First, a Kramers–Kronig transform exists of a B-spline curve, i.e., \( \varepsilon_1 \) can be found from a Kramers–Kronig transformation of \( \varepsilon_2 \). This reduces the number of fitting parameters by two and therefore also reduces the probability of correlation between them. Kramers–Kronig consistency requires that \( \varepsilon_2 \) goes smoothly to zero. This is ensured by choosing knots at appropriate locations outside the measured range. These outer knots should, therefore, also be communicated. Second, B-splines have a property known as convex hull: if all coefficients \( c_i \) are positive then the total curve is also positive. Since \( \varepsilon_2 \) can never be negative, all \( c_i \)'s should be positive.

By enforcing that the B-spline curve is Kramers–Kronig consistent and that all \( c_i \)'s are positive, a physical result for the parametrized dielectric function is ensured. Both conditions are enforced for all the B-spline parametrizations in this work.

3. The Cauchy model

Hydrogenated amorphous carbon thin films are semi-transparent, as is evident from the interference fringes in the long wavelength region in Fig. 3. The transparent part can be fitted by empirical Cauchy dispersion relation (3a) for the refractive index. The Cauchy dispersion relation by itself is not Kramers–Kronig consistent, since \( \varepsilon_2 \) is assumed to be zero. However, absorptions can be accounted for by adding relation (3b) to the dispersion relation

\[ n(E) = A + B E^2 + C E^4, \]

(3a)

\[ k(E) = D e^{F(E-E_{\text{edge}})}, \]

(3b)

with \( A, B, C, D, F \) and \( E_{\text{edge}} \) the fitting parameters and \( E_{\text{edge}} \) the band edge. Both relations together are hereafter referred to as the Cauchy model. Good estimates for thickness and roughness are found from this model. After determining the thickness and roughness from the transparent part of the data, the optical constants for the entire spectrum, including the absorbing part, can be found by an exact direct numerical inversion. This inversion is carried out wavelength by wavelength. Due to noise in the experimental data, it is not guaranteed that the dielectric function, resulting from the numerical inversion, is indeed Kramers–Kronig consistent.

4. Tauc-Lorentz oscillator

A Kramers–Kronig consistent parametrization of \( \varepsilon_2 \) by the TL oscillator is given by
Table I. Sample list.

<table>
<thead>
<tr>
<th>Section</th>
<th>Sample</th>
</tr>
</thead>
<tbody>
<tr>
<td>III A: Cauchy model</td>
<td>A</td>
</tr>
<tr>
<td>III B: The dielectric function in ambient air</td>
<td>A</td>
</tr>
<tr>
<td>III C: Growth rate and homogeneity</td>
<td>B</td>
</tr>
<tr>
<td>III D: TL oscillator</td>
<td>A</td>
</tr>
</tbody>
</table>

\[
e_2 = \begin{cases} 
  \frac{AE_0 \Gamma (E - E_g)^2}{E^2 (E^2 - E_0^2) + \Gamma^2 E^2}, & E \geq E_g \\
  0, & E < E_g 
\end{cases}
\]  

(4)

where \( E_g \) is the Tauc gap for amorphous materials and \( A = e^2 \nu / e_0 m_e \) is for the total number of electrons \( \nu \) with mass \( m_e \), \( e_0 \) the dielectric constant in vacuum, and elementary charge \( e \). The amplitude of the oscillator is \( A / \Gamma E_0 \), which has its maximum at \( E_0 \), and \( \Gamma \) represents the full width, half maximum.

5. Pole

In the last part of the experimental results, \( \text{ex situ} \) data will be analyzed by a TL oscillator in combination with a pole. This is a Lorentz oscillator without broadening

\[
e_{\text{pole}} = \frac{A}{E_{\text{pole}}^2 - E^2},
\]

(5)

with amplitude \( A \) at an energy position \( E_{\text{pole}} \). The pole, which affects only \( e_1 \) of the dielectric function, takes absorptions outside the measured range (see Sec. III D) into account. Because \( A \) and \( E_{\text{pole}} \) are susceptible to correlation, the global fit option of CompleteEASE is used. A global fit divides the parameter space into a grid of starting values for the fitting procedure. The parameters we used for the global fit itself are 20 iterations for each set of starting values and a limitation of the measured data to 100 datapoints. The set of starting values with the lowest \( x^2 \) is subsequently used for an extensive fit. Compared to the B-spline method, a global fit of a pole is computationally much more demanding and thus takes a longer time to complete. Despite the higher computational requirements of a pole, it does give a physical representation of absorptions outside the measured range.

III. EXPERIMENTAL RESULTS

The experimental results are elucidated by means of two samples (see Table I). Sample A is used for the \( \text{ex situ} \) results in Secs. III A, III B, and III D For the second sample, in Sec. III C, the data are obtained during deposition.

A. Cauchy model

\( \Psi \) and \( \Delta \), as shown in Fig. 3, are the results of a standard (\( \text{ex} \) \( \text{situ} \)) SE measurement of our \( a\)-C:H thin films. The transparent part of the film, characterized by interference fringes in \( \Psi \) and \( \Delta \), is dependent on the film thickness. This typically goes up to 2.4 eV for a thickness of around 1100 nm. An EMA layer of 50% bulk material and 50% voids is used for the roughness layer. The Cauchy model indicates that this particular sample has a film thickness and surface roughness of, respectively, 940 and 6 nm. The Cauchy dielectric function is used as a starting point for the B-spline parametrization in the section hereafter.

B. The dielectric function in ambient air

Before ascertaining the (in)homogeneity in depth of the hydrogenated amorphous carbon thin film as deposited by remote plasma deposition, the dielectric function in ambient air of the carbon film will first be determined. This is entirely possible by means of the B-spline model, which makes no assumptions about the physical properties of the deposited film itself.

Throughout the following steps, the roughness layer is included in every fit of the data. Although direct application of the B-spline model is possible, we first apply the Cauchy model to the transparent part of the measured data (see Sec. III A). Optical constants obtained by the Cauchy model are parametrized by B-splines. These B-splines are fitted directly to the experimental data by expanding the data range in steps of 0.5 eV from the band edge onwards to higher energies. If necessary, the number of knots can be decreased to smooth out the dielectric function from the previous step. As a last step, Kramers–Kronig consistency of the B-spline parametrization is enforced (see Sec. II B 2).

The dielectric function, expressed in \( e_1 \) and \( e_2 \), is shown in Fig. 4. Since the dielectric function is Kramers–Kronig consistent, only the spline curve for \( e_2 \) is tabulated (see Table II). The spline curve of \( e_2 \), for this particular sample, is defined by twelve knots and eight coefficients. Six knots are equally spaced between 1.24 and 6.50 eV and six knots are located outside this range at 0.64, 0.84, 1.04, 10, 20, and 21 eV. This means that eight B-spline coefficients are fitted: six for the knots in the measured range and two outside this range at 1.04 and 10 eV. The dielectric spectrum obtained via a wavelength-by-wavelength fit is also plotted for comparison.

![Figure 4](https://example.com/fig4.png)
further will reduce χ^2 even more. However, too many knots will only provide a better description of the noise and increase the — probability of — correlation between the coefficients. When using B-splines, a balance should be found between obtaining a low χ^2 value and the number of knots used to reach that value. The drop in χ^2 should be significant, compared to the additional number of knots. Therefore, it is better to provide the absolute minimum number of knots required to obtain a good fit. In addition to the χ^2 value, the correlation r between the fitting parameters should be taken into account. The correlation between the different parameters for this particular sample is shown in Table III. When the threshold for correlation versus no correlation between various parameters is set to r = 0.92, then there is no correlation between thickness, roughness, and the B-spline coefficients. The B-spline coefficients that fall well within the measured data range show the lowest correlation values. The outer B-spline coefficients (i.e., outside of the measured range), however, do show higher values for the correlation with the B-spline coefficients at the edge of the data range. This is to be expected, since only a few B-spline coefficients contribute to the spline curve for any given wavelength value and the outer B-spline coefficients have no measured data to be compared against. The B-spline dielectric function in Fig. 4 is compared with the dielectric function obtained from the wavelength-by-wavelength fit of Sec. III A. There is a significant overlap between the dielectric function obtained from the wavelength-by-wavelength fit and the B-spline parametrized dielectric function. This overlap between both parametrizations validates the use of the B-spline model for α-C:H films.

Together with the dielectric function, SE yields the film thickness and the roughness. With a roughness layer that is at most 1% of the total thickness, the deposited film can be considered smooth. The roughness obtained via SE is compared with an AFM roughness. An AFM (NT-MDT solver P47 with NSG 10 tips), operating in tapping mode to avoid damage to the sample scans a 2 × 2 μm² area with a resolution of 512 × 512 points. If all the measured heights follow a normal distribution, then the AFM roughness is defined as the standard deviation σ of the height distribution. The AFM and SE roughness are, respectively, 4.9 and 7.6 nm. The proportionality factor between the AFM and SE roughness is 1.55 for this α-C:H sample, whereas Kim et al.[23] found a proportionality factor of 2.1. The AFM roughness is in good agreement with the roughness as determined by SE. Cross-sectional SEM (model JEOL 7500FA) measurements (see Fig. 5) of another sample indicates a thickness comparable with a SE thickness of around 1100 nm on average. With the exception of scattered debris near the edge, which was caused by the cutting process, the SEM image (Fig. 6) also

<table>
<thead>
<tr>
<th>Roughness</th>
<th>Thickness</th>
<th>1.04</th>
<th>1.24</th>
<th>2.292</th>
<th>3.345</th>
<th>4.397</th>
<th>5.449</th>
<th>6.501</th>
<th>10.001</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>-0.374</td>
<td>0.137</td>
<td>-0.07</td>
<td>-0.402</td>
<td>-0.375</td>
<td>-0.076</td>
<td>-0.056</td>
<td>-0.101</td>
<td>0.437</td>
</tr>
<tr>
<td>-0.374</td>
<td>1</td>
<td>0.105</td>
<td>-0.124</td>
<td>-0.036</td>
<td>0.131</td>
<td>0.056</td>
<td>-0.048</td>
<td>0.263</td>
<td>-0.473</td>
</tr>
<tr>
<td>1</td>
<td>0.137</td>
<td>0.105</td>
<td>1</td>
<td>-0.852</td>
<td>0.34</td>
<td>-0.27</td>
<td>0.084</td>
<td>-0.041</td>
<td>-0.075</td>
</tr>
<tr>
<td>1.240</td>
<td>0.07</td>
<td>-0.124</td>
<td>-0.852</td>
<td>1</td>
<td>-0.551</td>
<td>0.38</td>
<td>-0.167</td>
<td>0.085</td>
<td>0.014</td>
</tr>
<tr>
<td>2.292</td>
<td>-0.402</td>
<td>-0.036</td>
<td>0.34</td>
<td>-0.551</td>
<td>1</td>
<td>-0.38</td>
<td>0.334</td>
<td>-0.132</td>
<td>0.142</td>
</tr>
<tr>
<td>3.345</td>
<td>-0.375</td>
<td>0.131</td>
<td>-0.27</td>
<td>0.38</td>
<td>-0.38</td>
<td>1</td>
<td>-0.478</td>
<td>0.273</td>
<td>-0.009</td>
</tr>
<tr>
<td>4.397</td>
<td>-0.076</td>
<td>0.056</td>
<td>0.084</td>
<td>-0.167</td>
<td>0.334</td>
<td>-0.478</td>
<td>1</td>
<td>-0.571</td>
<td>0.472</td>
</tr>
<tr>
<td>5.449</td>
<td>-0.056</td>
<td>-0.048</td>
<td>-0.041</td>
<td>0.085</td>
<td>-0.132</td>
<td>0.273</td>
<td>-0.571</td>
<td>1</td>
<td>-0.525</td>
</tr>
<tr>
<td>6.501</td>
<td>-0.101</td>
<td>0.263</td>
<td>-0.075</td>
<td>0.014</td>
<td>0.142</td>
<td>-0.009</td>
<td>0.472</td>
<td>-0.525</td>
<td>1</td>
</tr>
<tr>
<td>10.001</td>
<td>0.437</td>
<td>-0.473</td>
<td>0.16</td>
<td>0.072</td>
<td>-0.25</td>
<td>-0.194</td>
<td>-0.415</td>
<td>0.291</td>
<td>-0.894</td>
</tr>
</tbody>
</table>

FIG. 5. Cross-sectional SEM image of an α-C:H sample, with magnification 75 000×. Crack lines, due to breaking the sample, are visible.

TABLE III. Correlation values between the thickness, roughness and the eight B-spline coefficients, for α-C:H in ambient air (see also Fig. 4).
shows a uniform, smooth film. The stack of layers visible in the cross section is a validation of the use of our multilayered model.

**C. Growth rate and homogeneity in depth of the a-C:H film**

Assuming linear growth, only the deposition time and thickness after deposition are necessary to calculate a growth rate. From *in situ* SE measurements during deposition, however, also the linearity of the growth rate can be investigated. Also the homogeneity in depth of the deposited film is investigated by comparing the dielectric functions for every measured thickness. If the deposited carbon layer is homogeneous, then the dielectric constants should stay the same throughout the deposition process. *In situ* SE data are therefore gathered with the highest time resolution possible (4.8 Hz). By manually shutting down the plasma source after 32 s in this experiment, the shutter did not block the light path of the ellipsometer and the measurement could continue after the end of the deposition. The *in situ* data have a different wavelength range compared to the *ex situ* data. Therefore, a different set of knots is used (see Table IV for the knot positions). At the moment, limitations in computing power, necessary to calculate the dielectric function for every datapoint, hinder us from performing real time *in situ* analysis. The *in situ* data are therefore analyzed postdeposition. The analysis of these time dependent data occurs stepwise, whereby the dielectric function of the previous datapoint acts as the starting point for the next datapoint. Since the film undergoes nucleation during the initial stages of growth, thickness, roughness, and optical constants are correlated during this stage. Therefore, the data are analyzed backward in time. This allows for a good initial determination of the dielectric function.

Figure 7 shows the total thickness evolution, i.e., $d_{\text{total}} = d + \frac{1}{2} d_{\text{roughness}}$, as a function of deposition time. The void fraction is kept constant at 50%. The growth rate, here defined as the slope of a linear fit of the total thickness, is $35.7 \pm 0.1$ nm/s. Although the acquisition rate is set to 1 rev/measurement, the actual data are averaged over 4 revs/measurement. With a growth rate of 35.7 nm/s, the film grows with about 7.5 nm during one acquisition interval.

Also the evolution of the $\varepsilon_2$-spectra is plotted (see Fig. 8). These spectra are averaged over the entire deposition interval. Since the B-spline coefficients in the first 4 s (i.e., nucleation) are correlated with the thickness and roughness, these coefficients are excluded from the average. The average of the B-spline coefficients for the $\varepsilon_2$-spectra is tabulated in Table IV. The deviations in $\varepsilon_2$ at energies above 4 eV are

<table>
<thead>
<tr>
<th>Knot position ($t_i$)</th>
<th>B-spline coefficient ($c_i$)</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.152</td>
<td>n/a</td>
</tr>
<tr>
<td>0.352</td>
<td>n/a</td>
</tr>
<tr>
<td>0.552</td>
<td>0.066 00</td>
</tr>
<tr>
<td>0.752</td>
<td>0.001 01</td>
</tr>
<tr>
<td>1.828</td>
<td>0.626 02</td>
</tr>
<tr>
<td>2.903</td>
<td>1.877 71</td>
</tr>
<tr>
<td>3.979</td>
<td>2.229 73</td>
</tr>
<tr>
<td>5.054</td>
<td>1.858 46</td>
</tr>
<tr>
<td>10.054</td>
<td>1.017 14</td>
</tr>
<tr>
<td>20.054</td>
<td>n/a</td>
</tr>
<tr>
<td>21.054</td>
<td>n/a</td>
</tr>
</tbody>
</table>

**FIG. 6.** Cross-sectional SEM image of an a-C:H sample, with magnification 16 000×. The film is uniform and smooth. The large features on the surface are debris from breaking the sample.

**FIG. 7.** (Color online) Evolution in total thickness (i.e., thickness plus half of the roughness) during deposition. After 32 s, the plasma is stopped.

**FIG. 8.** (Color online) $\varepsilon_2$ spectra during deposition, as determined by the B-spline model. The deviation from the average (dashed line) at higher energies is attributed to relatively larger measurement errors in this energy range.
attributed to relatively larger measurement errors in $\Psi$ and $\Delta$ for this energy range. With the exception of these deviations above 4 eV, $\varepsilon_2$ remains constant during deposition. The deposited film is therefore considered homogeneous in depth.

Since the deposited a-C:H thin film is homogeneous in depth, the dielectric constants — as parametrized by the average B-spline coefficients — can be fixed for every measurement point. Refitting the data with fixed dielectric constants yields, for a second time, the evolution in thickness and roughness. The film thickness, averaged over the time interval from 33 to 48 s, is this time about 1.2% higher. The roughness evolution during deposition is shown in Fig. 9. In the first 3 s of deposition the roughness reaches a maximum of 4 nm, indicating the nucleation process at the start of the deposition. After nucleation, the roughness stays roughly constant at nearly 2 nm.

D. TL oscillator

The mathematically accurate description of the dielectric function, obtained via the B-spline model, is used as a starting point for the parametrization of the bulk layer by one or more TL oscillators. Such a parametrization provides a physical model for the homogeneous carbon bulk layer. Both the SEM and AFM measurements agree well with the layer structure as described for the B-spline model (see Sec. III B). This layer structure is, therefore, reusable in a physics-based model.

Carbon is a band gap material, with a wide gap between the $\sigma$ valence band and $\sigma^*$ conduction band and a smaller band gap between the valence and conduction bands of $\pi$ and $\pi^*$, respectively. The $\pi-\pi^*$ and $\sigma-\sigma^*$ electronic transitions dominate the dielectric function of a-C:H (see Fig. 1). Each transition is commonly modeled by a TL oscillator. Although mixing of the $\sigma$ and $\pi$ bands occurs in the intermediate energy region between 5 and 8 eV, the resulting $\pi-\pi^*$ and $\sigma-\sigma^*$ electronic transitions can be neglected.

The same sample as in Sec. III B is reanalyzed. The measurement range of the ex situ data is limited by the equipment to a maximum energy of 6.5 eV, which limits us to a single TL oscillator for the $\pi-\pi^*$ transition. However, absorptions outside the measured range, in particular, due to the $\sigma-\sigma^*$ electronic transition, can be partially taken into account by adding a constant offset to $\varepsilon_2(\omega)$. It is obvious from the fit in Fig. 10, which has a $\chi^2$ of 12.7, that adding a constant offset to $\varepsilon_1$ is only sufficient up to 4 eV. This can be improved by replacing the offset with a pole, which reduces the $\chi^2$ to 5.8. In both variants of this model, the thickness (933 nm) and roughness (6.7 nm) of the layer were first determined by the B-spline model and subsequently fixed.

The dielectric function of the carbon layer, as determined by the B-spline model, is then parametrized by a combination of a TL oscillator and an offset $\varepsilon_\infty$. The TL parameters obtained in this fit are subsequently used as the starting values for a TL oscillator whereby the offset is replaced with a pole. The parameter space of the global fit (see also Sec. II B 5) for $E_{\text{pole}}$ goes from 9 to 20 eV in 24 intervals. The range for $\Delta$ is 0–1000 eV divided in 20 intervals. The results of the fit are shown in Table V. In contrast with the B-spline model, the TL oscillator with a pole also provides physical information about the carbon film. The TL oscillator describes the $\pi-\pi^*$ transition, whereas the pole indicates the location of the $\sigma-\sigma^*$ electronic transition. The B-spline model is thus used as a stepping stone to a physics-based model, from which additional information about the bulk layer can be extracted.

IV. DISCUSSION

We have shown that hydrogenated amorphous carbon thin films, as deposited by remote plasma deposition, can be...
represented by a fairly simple and straightforward set of layers. The first two layers are the substrate itself, consisting of a (semi-infinite) Si wafer of 1 mm thick and a native oxide layer of 1.6 nm. Layer 3 is the homogenous bulk carbon layer. The fourth and last layer is the roughness of the film, which is described by Bruggeman’s EMA with a mix of 50% voids and 50% bulk material. We have verified this layer structure by cross-sectional SEM and AFM measurements.

The B-spline model, which was instrumental in determining the film thickness and roughness, also provides an excellent method to determine the dielectric function of the carbon film throughout the deposition, although only for postdeposition analysis. The average of the dielectric function shows a tail below 0.7 eV, outside of the measured wavelength range. Although the B-spline parametrization of \( \varepsilon_2 \) goes smoothly to zero within the measured wavelength range, it accomplishes this at the expense of the wavelengths outside of the measured range. This tail, therefore, an artifact of the mathematical representation of \( \varepsilon_2 \). The B-spline parametrization can only accurately represent the available data. Extrapolation to wavelengths outside of the measured range is thus not possible, as illustrated by the tail below 0.7 eV. The evolution of the dielectric function was used to establish the homogeneity in depth of the a-C:H layer. Since the film is homogeneous, the average dielectric constants should be used when determining the thickness evolution during growth. However, the difference in thickness is about 1.2% when the dielectric constants are included in the fit. Fixing the dielectric constants to the average value is, therefore, not necessary to get a good determination of the thickness evolution. Previous studies on the growth of a-C:H under similar deposition conditions on the same setup used infrared interferometry to determine the growth rate.\(^{25,26}\) The rate found by Gielen et al. is comparable to what we found by means of \textit{in situ} SE. Our analysis shows a deposition rate of about 36 nm/s. Even though the integrated thickness during one acquisition interval is 7.5 nm, the \textit{in situ} data acquisition was fast enough to observe the nucleation stage of the film, which lasted no more than 3 s. With a roughness layer thinner than 1% of the total thickness, the film can be considered smooth.

The B-spline model served as a stepping stone for a physics-based model. In this model, the dielectric function is parametrized by means of a TL oscillator in combination with either an offset or a pole. Although the TL oscillator with a pole is based on a physical model, it is computationally slower than the mathematical approach via the B-spline model. Moreover, the B-spline model is less susceptible to correlation between the different fitting parameters. Therefore, if only the dielectric spectrum in the measured wavelength range and a fast and accurate thickness determination are of primary concern to the experimentalist, the B-spline model is recommended for hydrogenated amorphous carbon thin films. Application of the B-spline model is, however, not limited to hydrogenated amorphous carbon thin films.\(^{14}\) The B-spline model is also suitable for \textit{in situ} monitoring the growth of, e.g., \( \mu \)-c-silicon. The change from \( \alpha \)-silicon to \( \mu \)-c-silicon has a clear distinction in the dielectric function. It is envisaged that in due time, due to the ever increasing CPU computing speed, the B-spline method can be used to monitor real time the changes in the dielectric function of a-C:H \textit{in situ}, during deposition or etching.

V. CONCLUSION

We have shown the versatility of the B-spline model as a tool in the analysis of a-C:H thin films. We concluded that the deposited film is smooth and homogeneous in depth. The B-spline model can serve as a stepping stone to more physical models such as the TL oscillator. The expanding thermal plasma deposition technique has a high and constant deposition rate. In all, the B-spline model is an accurate and fast method to determine thickness, roughness, and dielectric constants of — as has been shown — hydrogenated amorphous carbon thin films, both for \textit{ex situ} and \textit{in situ} measurements.

ACKNOWLEDGMENTS

We would like to thank W. Keuning for the SEM measurements and V. Vandalon for the AFM measurements. We also greatly appreciate the skillful technical assistance of M. J. F. van de Sande, J. J. A. Zeebregts, and H. M. M. de Jong. This work is part of the research program of the Dutch Foundation for Fundamental Research on Matter (FOM-TFF). It is also supported by the European Communities under the contract of Association between EURATOM and FOM and carried out within the framework of the European Fusion Programme.

26A change in one of the coefficients affects the next four segments, as can be seen from the recursive formula.


