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Critical reexamination of resonant soft x-ray Bragg forbidden reflections in magnetite

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Magnetite, Fe_3O_4 , displays a highly complex low-temperature crystal structure that may be charge and orbitally ordered. Many of the recent experimental claims of such ordering rely on resonant soft x-ray diffraction at the oxygen K and iron L edges. We have reexamined this system and undertaken soft x-ray diffraction experiments on a high-quality single crystal. Contrary to previous claims in the literature, we show that the intensity observed at the Bragg forbidden $(00\frac{1}{2})_c$ reflection can be explained purely in terms of the low-temperature structural displacements around the resonant atoms. This does not necessarily mean that magnetite is not charge or orbitally ordered but rather that the present sensitivity of resonant soft x-ray experiments does not allow conclusive demonstration of such ordering.

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In many transition-metal oxides, the spatial localization of electrons on certain sites, the so-called charge ordering (CO), has been used to explain some of their more intriguing ground-state properties. For example, charge ordering has been invoked to describe phase transitions in some magnetoresistive manganites,¹ and the dynamic fluctuations of charge-ordered stripes² have been proposed as a mechanism of high-temperature superconductivity.³ Magnetite, Fe_3O_4 , was the first material in which such a charge ordering transition was proposed, in connection with the metal-insulator transition discovered by Verwey,⁴ and it has long been interpreted as the classic example of mixed-valence compound with formula unit $\text{Fe}^{3+}[\text{Fe}^{2+}\text{Fe}^{3+}]\text{O}_4$ (Refs. 4 and 5). In this interpretation, magnetite, which at room temperature crystallizes into the cubic inverted spinel structure $AB_2\text{O}_4$, with space group $Fd\bar{3}m$, has formally Fe^{3+} ions at the tetrahedral A sites and formally Fe^{2+} and Fe^{3+} ions at the octahedral B sites. Unfortunately, this simple picture is deceptive as the crystal structure below T_V is extraordinarily complicated: the most recently reported structure⁶ consists of a complex $P2/c$ monoclinic cell containing 56 atoms in which the A and B Fe ions are split between two and six inequivalent sites, respectively. Based on the $P2/c$ structure, local spin density approximation (LSDA+ U) band-structure calculations have reported both CO (0.16 electrons) and an associated t_{2g} orbital ordering (OO) on the octahedral sublattice.^{7,8} Seemingly arguing against this, however, are the results of resonant x-ray scattering (RXS) experiments at the iron K edge, which were interpreted as providing evidence either against any charge ordering^{9–11} or in favor of a 0.12 electrons charge disproportionation¹² between the formally Fe^{2+} and Fe^{3+} sites than those predicted in Refs. 7 and 8. Very recently two further papers have appeared in which soft-x-ray diffraction measurements at the O K edge¹³ and the Fe L edges¹⁴ were interpreted as providing evidence for both charge and orbital

orderings. We note that in the monoclinic cell the iron B sites are no longer equivalent by symmetry and there is therefore no requirement that they have the same charge density surrounding the atomic site. It is thus most likely that they will not have the same charge density. The question is therefore: what is the smallest charge difference, about which one would reasonably claimed that the material is charge ordered?

Resonant x-ray scattering occurs when a photon excites a core electron into an excited state and is subsequently re-emitted when the electron and core hole combine.¹⁵ On resonance the x-ray scattering amplitude is anisotropic and is sensitive to the anisotropic charge distribution of the resonating ion. The anisotropic charge distribution can be intrinsic to the scattering ion due to orbital occupation or can be intrinsic to the lattice as in the case of Templeton-Templeton scattering.^{16–18} The characteristic of Templeton-Templeton scattering is that a reflection which is Bragg-forbidden because of a compound symmetry operation, such as a glide plane or screw axis, becomes allowed when the incident photon energy is tuned to a resonance. On resonance the x-rays are sensitive to the quadrupolar term in the charge distribution, of the resonating atom, Q , and the difference between the two electric quadrupole moments, related by the symmetry operation, sum to zero and a resonant peak, is observed arising from the crystal structure.

We have chosen to revisit magnetite and report here on resonant soft x-ray experiments that confirm the resonant enhancement of the $(00\frac{1}{2})_c$ reflection at both the oxygen K and iron L_3 edges. However, we have carried out a careful analysis of this superlattice reflection focusing on a detailed investigation of the effects of the distorted crystal structure below the Verwey transition, *without* invoking any charge or orbital ordering. We find that we can model our data well by considering only Templeton-Templeton scattering arising due to

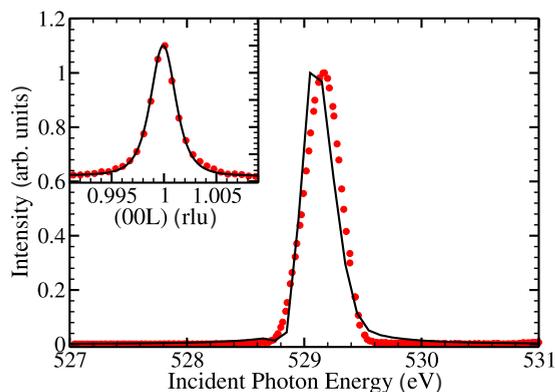


FIG. 1. (Color online) The incident photon energy dependence of the $(001)_o$ reflection in Fe_3O_4 close to the oxygen K edge (red circles). The solid black line represents *ab initio* calculations of the scattered intensity assuming structural distortions and no charge nor orbital order. In the insert, a scan along the $[001]_o$ direction through the $(001)_o$ reflection is shown (red circles); the solid line is a fit to a Lorentzian-squared line shape.

the structural distortions below the Verwey transition, without the need to resort to charge order or orbital order. This is contrary to the claims of Refs. 13 and 14.

The experiments were conducted on high-quality synthetic magnetite crystals prepared in an arc-image furnace using the floating-zone technique. The purity of the sample was verified by heat-capacity measurements, which gave a maximum heat-capacity value of 120.6 K and an entropy change of 5.77 J K^{-1} at the Verwey transition. These results give for $\text{Fe}_{3-\delta}\text{O}_4$ a value $\delta=0.0002$ showing that the stoichiometry of our sample is very close to the ideal case. Soft x-ray diffraction experiments were conducted on the ID08 beamline at the ESRF, Grenoble, France. In what follows we index the sample in the approximate low-temperature $Pmca$ orthorhombic structure (No. 57) with lattice parameters $a=5.944 \text{ \AA}$, $b=5.925 \text{ \AA}$, and $c=16.775 \text{ \AA}$.⁶ This structure is related to the $P2/c$ (No. 13) structure by only a slight monoclinic distortion ($\beta=90.2363^\circ$). In this orthorhombic setting, the cubic $(00\frac{1}{2})_c$ reflection becomes the orthorhombic $(001)_o$. The sample of Fe_3O_4 was cut with a $[001]_o$ surface normal and polished with $0.1 \mu\text{m}$ diamond paste. It was then mounted on a SmCo magnet, providing a field at the sample surface of $\approx 0.3 \text{ T}$, parallel to the surface normal. This field defines a unique c axis so that on cooling through the transition the number of crystallographic domains is minimized. The sample was then cooled to a base temperature of 30 K, and a resonant signal was observed at the $(001)_o$ position in reciprocal space in the vicinity of the iron $L_{2,3}$ and oxygen K edges.

Figure 1 shows an incident photon energy scan at constant wave vector of $(001)_o$ as the is tuned in the vicinity of the oxygen K edge. The scattering is observed to peak at 529.1 eV, about 10 eV below the main oxygen K edge. The insert shows a scan along the $[001]_o$ direction through the $(001)_o$ reflection at $E_i=529.1 \text{ eV}$, with a fit to a Lorentzian-squared line shape. The correlation length obtained from this fit is $>3000 \text{ \AA}$. This represents a lower bound on the penetration depth of the x-rays, and thus this value indicates that the

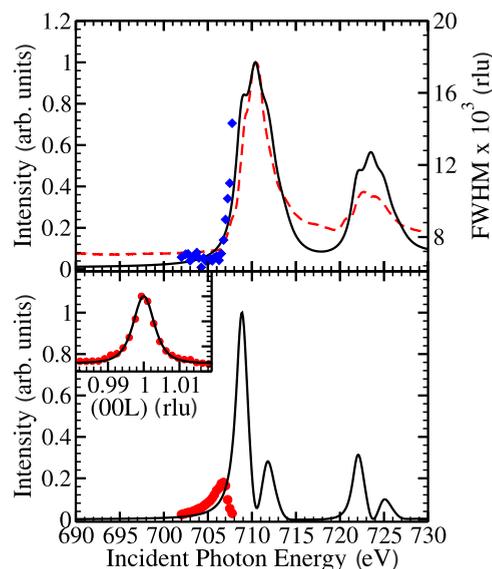


FIG. 2. (Color online) Top panel: measured total electron yield (x-ray absorption spectrum) at 27 K (red dashed line) and calculated absorption spectrum (solid black line) through the Fe $L_{2,3}$ edges. Bottom panel: incident photon energy dependence of the integrated intensity of the $(001)_o$ reflection close to the Fe L -edges (red circles). The solid black line shows *ab initio* calculations of the resonant scattering (see text). The inset represents a scan along the $[001]_o$ direction at an energy corresponding to the maximum in the signal. The variation in the longitudinal width of the $(001)_o$ reflection with incident energy is included in the top panel for comparison with the experimental and calculated absorption (blue diamonds).

resonant signal is not surface sensitive even at the maximum of the resonance. The $(001)_o$ reflection was also visible in the vicinity of the iron L_3 edge (Fig. 2). The bottom panel of Fig. 2 shows the incident photon energy dependence of the integrated intensity of the $(001)_o$ reflection. The experimental signal is only visible at the L_3 threshold, with a maximum at an energy of 706.5 eV and is found to be suppressed above 708 eV. Such behavior arises from the very large self-absorption caused by the strong Fe L_3 resonance, leading to a total loss in the observed signal. The width of the diffraction peak as a function of energy as shown in the top panel of Fig. 2 confirms this. The peak width is broader than that found at the oxygen K edge and tracks the calculated absorption (dashed line) indicating that the change in width arises from the increased absorption and consequently reduced penetration depth. Finally, Fig. 3 shows the temperature dependence of the integrated intensity of the $(001)_o$ reflection measured at both the iron L_3 and oxygen K edges. The data were collected by performing rocking scans of the sample angle, θ , at each temperature. The signal at both edges was found to be virtually constant up until a temperature of $\approx 125 \text{ K}$ above which no intensity is observed.

We now turn to our resonant scattering simulation. We have used the FDMNES program¹⁹ in the multiple-scattering mode. (The results of these simulations are shown in Figs. 1 and 2 for the oxygen K and iron L edges respectively.) In order to calibrate to the experimentally obtained data with the FDMNES simulations, the calculated absorption was com-

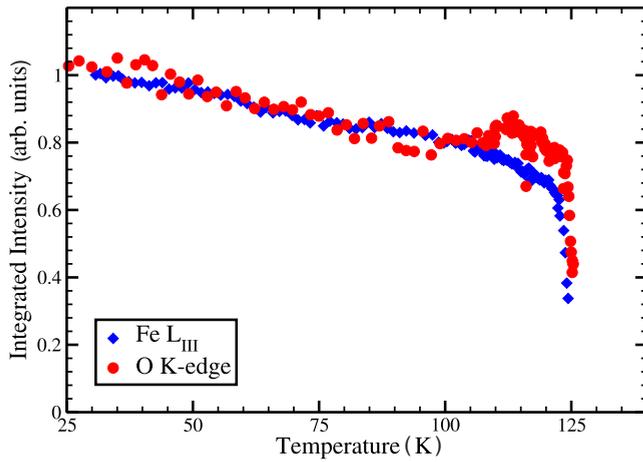


FIG. 3. (Color online) Temperature dependence of the (001) reflection of Fe_3O_4 at the O K edge (red circles) and the iron L_3 edge (blue diamonds).

pared with the sample absorption measured by total electron yield at the oxygen K edge. Our simulation reproduces well the main experimental features, including the energy gap of about 10 eV between the RXS signal and the main oxygen absorption edge, as well as the energy width of the peak.

In this specific case, there are eight inequivalent oxygen sites of $4d$ Wyckoff symmetry in the $Pmca$ space group. Considering only the oxygen atoms which dominate at this energy, the structure factor of the $(001)_o$ reflection is

$$S_{(001)_o} = \sum_{j=1, \dots, 8} 2f_j(1 - \hat{m}_y) \cos(2\pi w_j), \quad (1)$$

where f_j is the atomic scattering amplitude, ($j=1, \dots, 8$ labels the inequivalent sites), w_j is the fractional coordinate of the j th oxygen atom in the c direction, and \hat{m}_y is the mirror plane in the b direction of the $Pmca$ setting.⁶ Using the local mirror symmetry \hat{m}_x of the $4d$ sites, we find that $f_j \propto Q_{yz}^j$, the electric quadrupole matrix element. In the monoclinic $P2_1/c$ setting, the \hat{m}_x symmetry is lost and a further contribution $f_j \propto Q_{xy}^j$ appears. This can be shown to be negligible since it is proportional to the small angular distortion, $\beta \neq 90^\circ$, from the orthorhombic $Pmca$ structure. Upon evaluating the structure factor S we can conclude that almost all the scattered intensity comes from the sum of the quadrupoles Q_{yz} at oxygen sites O_1 and O_2 only. That is $S_{(001)_o} \approx Q_{yz}^{O_1} + Q_{yz}^{O_2}$ in the high-temperature phase $Q_{yz}^{O_1} = -Q_{yz}^{O_2}$ due to the \hat{C}_{2z} screw axis of the high-temperature $Fd\bar{3}m$ space group, and therefore their sum is zero. Below the Verwey transition, the unequal atomic displacements of the O_1 and O_2 oxygen sites from their high-temperature positions give a finite signal even if $Q_{yz}^{O_1} = -Q_{yz}^{O_2}$. However, this signal is tiny because of the very small displacement, and has an expected amplitude of $\approx 10^{-4} \times |Q_{yz}^{O_1}|$. In contrast, a much bigger amplitude might be expected if the surrounding iron tetrahedra are distorted making $Q_{yz}^{O_1} \neq -Q_{yz}^{O_2}$.

To investigate which of these contributes to be the most significant we have performed several numerical calculations in which the tetrahedral or octahedrally coordinated iron at-

oms, and/or the oxygens were in turn placed in their high-temperature positions, with the rest of the cluster held in their low-temperature positions. By this method, we found that the main contribution to the signal comes from the O_2 position ($\sim 70\%$ of the total): in fact $Q_{yz}^{O_1}$ does not change much when the iron sites move from the high-temperature to the low-temperature positions while $Q_{yz}^{O_2}$ varies by about 100%. This dominant change is due in particular to the displacement of the octahedral iron atoms surrounding the O_2 cite. The iron sites belong to the Fe $B3$ sites, in the notation of Ref. 20, and undergo the strongest distortion when passing from the high-temperature to the low-temperature phase: the $\text{Fe}_{B3}-\text{O}_2$ distance changes from 2.06 to 1.96 Å, about a 5% contraction. Therefore the O K edge signal is mainly determined by the hybridization of $2p$ oxygen orbitals at O_2 sites with $3d$ iron orbitals belonging to octahedral Fe $B3$ sites.

The fact that we can explain the signal through these atomic displacements contradicts the interpretation of the signal at $(001)_o$ in Ref. 13 that explicitly excluded a structural origin to the $(001)_o$ reflection at the oxygen K edge. In particular, the arguments made by the authors of Ref. 13, for the assignment of their signal to charge and orbital orders due to its polarization dependence, are reproduced in our calculations based solely on structural distortions. This makes clear that invoking charge or orbital ordering to explain the detection of this superlattice reflection is unnecessary and potentially misleading.

This same procedure was then used to evaluate the resonant signal of the $(001)_o$ reflection at Fe $L_{2,3d}$ -edges, as shown in Fig. 2. The calibration between our experimental data and the FDMNES simulation was set by comparison of the calculated absorption with the absorption as measured by total electron yield.

We can repeat the same analysis as performed at the oxygen K edge for the data at the iron $L_{2,3}$ edges. The results of the FDMNES simulation are shown in Fig. 2. In the top panel a comparison between the measured and calculated absorptions is shown, while the bottom panel shows the comparison between the integrated intensity of the $(001)_o$ reflection as a function of incident photon energy and our simulation.

In the $Pmca$ setting there are six inequivalent groups of iron atoms, with two groups of tetrahedral iron sites ($A1$ and $A2$, following the notation of Ref. 20), and four groups of octahedral iron sites ($B1$, $B2$, $B3$, and $B4$), each group containing four iron atoms. $A1$, $A2$, $B3$, and $B4$ sites have a local \hat{m}_x symmetry ($4d$ Wyckoff site) so that the same considerations discussed above for the structure factor at the oxygen K edge are still valid. The only signal that can be measured for these ions is due to the \hat{Q}_{yz}^j , this time projected on the corresponding iron sites j . The $B2$ site has a \hat{C}_{2y} local symmetry ($4c$ Wyckoff site) and the two groups of two ions that contribute in antiphase at the $(001)_o$ are related by inversion symmetry, so their total contribution equal to zero. Finally the $B1$ sites, with local inversion symmetry ($4b$ Wyckoff site), also contribute with the quadrupole component \hat{Q}_{yz}^j . When we numerically compare the separate contributions of the $A1$, $A2$, $B1$, $B3$, and $B4$ sites, we find that the intensity from the $B1$ site is smaller by a factor of 500 relative to the contribution from the $B4$ site. We can conclude therefore that

the dominant contribution of the intense low-energy peak seen in the calculations at 708.5 eV is from the iron atoms sitting at the *B4* sites. The tetrahedral *A* sites and the octahedral *B3* site contribute mainly to the smaller shoulder at higher energy, and the fact that they have opposite amplitudes gives rise to the local minimum at 710 eV. Therefore all the experimentally detected signal comes from the *B4* sites (nominally Fe^{2+}). This is in keeping with some previous results obtained with rather different approaches which concluded that the low-energy part of L_3 spectrum (around 707 eV) is mainly determined by the t_{2g} states of the nominally divalent iron ions.^{7,21} However there is an important difference between the work presented here and atomic-multiplet-based calculations such as those reported in Refs. 14 and 21. By focusing on the orbital occupancy in a ionic model, these latter authors have neglected the important structural differences that exist between sites such as *B1* and *B4* (that are otherwise equivalent where formal charge is concerned, both being formally Fe^{2+}). The fact that we calculated their relative contribution to the total signal in the ratio 1:500 proves such a difference.

The similar temperature dependence shown in Fig. 3 can now be explained as a natural consequence of the signals at the oxygen *K* edge and iron *L* edges both measuring the same order parameter. As discussed above we argue that this order parameter is structural distortions associated with the structural phase transition from the $Fd\bar{3}m$ high-temperature structure to the low-temperature $P2_1/c$ structure.

In conclusion, we have shown that the $(001)_o$ reflection of Fe_3O_4 is sensitive to the local displacements around the resonant ion at both the oxygen *K* and iron *L* edges. The electronic anisotropy arising from the crystal distortions are sufficient to explain the origin of the scattered signals. It is clear that further work is required to ascertain the ratio in quantitative terms between any signal from charge or orbital ordering and the contribution from structural effects. It is clear that further studies will require more definitive structural information that is still unknown. At the oxygen *K* edge, the signal is determined by the hybridization of O $2p$ orbitals of the four O_2 atoms with the *B3* Fe $3d$ orbitals (nominally Fe^{3+}). At the iron L_3 edge, in contrast, the resonant x-ray scattering is mainly sensitive to the contribution of Fe $3d$ orbitals from *B4* sites (nominally Fe^{2+}).

We believe that our experiments and their analysis, while not contributing directly to the fashionable issue of charge ordering and orbital ordering, will help nonetheless understanding the structural transition of Fe_3O_4 below the Verwey temperature, at the same time indicating how a proper analysis of x-ray resonant scattering should, in our opinion, be conducted.

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